



# Luminescent and gas sensor properties of the $\text{ZrO}_2\text{:Hhpa:Eu}^{3+}$ Hybrid Compound

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## ABSTRACT

Luminescent complexes of rare earth have been extensively studied in different fields of research improving the industrial development of these materials for their utilization as electroluminescent displays, catalysts and biomaterials. The interest in rare earth compounds having aromatic ambident ligands has increased since it was found that they can act as light collectors (antenna effect), thus bypassing the weak absorption of the lanthanides and resulting in highly efficient luminophors. In this work  $\text{ZrO}_2$  was prepared by the hydrothermal microwave method (HMM) at 140 °C for 32 min. The hybrid materials was obtained using the previous  $\text{ZrO}_2$  dispersed in distilled  $\text{H}_2\text{O}$ , to which was added  $\text{Eu}^{3+}$  and the 3-hydroxypicolinamide (Hhpa), and then stirred for about 3 h. The materials were analyzed by X-ray diffraction (XRD) and Transmission electron microscopy (TEM). The photoluminescent properties were investigated through their excitation and emission spectra, and gas sensor studies were also performed. It was observed the presence of completely crystalline powders for all materials, and XRD peaks were indexed to the tetragonal phase of  $\text{ZrO}_2$  (JCPDS 50-1089). TEM micrographs reveal a spherical nanostructure pattern presenting an average diameter of 4 nm. Excitation and emission spectra revealed the characteristic peaks of the  $\text{Eu}^{3+}$  transitions. The studies of these powders as gas sensors showed a significative change in the electric resistance for the hybrid material.

## 1. Introduction

Rare earth luminescent complexes have been utilized in different fields in the last decade. These materials can be used in the fabrication of electroluminescent displays, fluorescent probes, catalysts, as well as biosensors and can be a promising candidate as gas sensors. The luminescent properties of  $\text{Eu}^{3+}$  complexes are very interesting due to their thin emission bands and their long lifetime.  $\text{Eu}^{3+}$  ion presents intraconfigurational  $4f-4f$  transitions which are Laporte forbidden and consequently presents low absorption intensity. The presence of organic ligands, however, can intensify the ability of  $\text{Eu}^{3+}$  ions emission through the ligand absorption and transfer part of this energy in an energy transfer mechanism called as antenna effect. This phenomenon resulted in various researches and different ligands were used to this purpose. These organic-inorganic hybrid materials have attracted great interest because of their potential applications in different aspects of technology like optical materials or laser systems [1–4].

Nanostructures of zirconium oxide,  $\text{ZrO}_2$ , have been explored due to their technological importance based on their high hardness, elevated

refractive index, optical transparence, chemical stability, low conductivity, high coefficient of thermal expansion, and high resistance to the corrosion [5,6]. These properties provide to  $\text{ZrO}_2$  different application in the field of optic [7], solid state electrolytes, gas sensors, and so on [8]. The application of  $\text{ZrO}_2$  in the photonic field is already know due to their high performance in wave guide and their efficient thermoluminescent response [9,10]. The properties of  $\text{ZrO}_2$  can be influenced by the preparation method, being possible to obtain metastable phases depending on the methodology employed. Microcrystalline  $\text{ZrO}_2$  is obtained when prepared at atmospheric conditions and it is obtained as amorphous, monoclinic, tetragonal, and cubic polymorphic phases [11,12]. So,  $\text{ZrO}_2$  can be present as three different crystalline structures P21/c monoclinic (below 1170 °C), P42/nmc (137) tetragonal (between 1170 and 2370 °C) and Fm3m (225) cubic (up to 2370 °C) [13].

The monoclinic phase (A) is less symmetric than the tetragonal one (B), and is more stable at room temperature. This phase is transformed in the tetragonal one at 1170 °C, but at 2370 °C the cubic phase (C) is formed [14]. The phases obtained at higher temperatures are unstable

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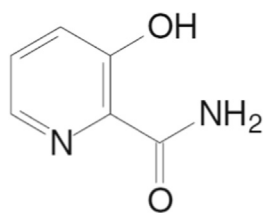


Fig. 1. Molecular structure of the 3-hydroxypicolinamide (Hhpa) ligand.

at room temperature, but these phases are more important for technological applications than the monoclinic one. So, various bivalent and trivalent cationic species like  $Mg^{2+}$ ,  $Ca^{2+}$ , and  $Y^{3+}$  are incorporated to the  $ZrO_2$  promoting the obtention of the cubic and tetragonal phases stables at room temperature [15].

Some works reported the doping of  $ZrO_2$  with different rare earths like  $Er^{3+}$ ,  $Eu^{3+}$ , and  $Sm^{3+}$  [16–18]. However, few of them focused on the photoluminescent properties of  $ZrO_2$  doped with  $Eu^{3+}$  [19–21]. The spectroscopic and magnetic properties of the rare earth ions are responsible for innumerable researches based on these elements. The luminescent spectra of the  $Eu^{3+}$  present narrow lines of excitation and emission, due to the electronic transitions inside the unfilled  $4f$  shell. Also because of the unfilled shell, the ion presents a great number of energy levels, given rise to emission lines that appears in the luminescence spectra from ultraviolet to the infrared, and depends on the kind of matrix they are incorporated. The  $4f$  orbitals are protected from the environment by  $5s$  and  $5p$  ones, so there are weak effects from the ligands field. The  $f-f$  transitions are forbidden by Laporte and spin rules, but these rules are broken when the symmetry around this ion is low [22]. The  $Eu^{3+}$  is the most studied from the rare earth ions due to its special characteristics. In general, the  $Eu^{3+}$  ion presents a more intense luminescence [23]. The characteristic red emission of  $Eu^{3+}$  is due to the transitions from the excited  ${}^5D_0$  level to the fundamental ones  ${}^7F_J$  ( $J = 0-6$ ) of its  $4f^6$  configuration. These characteristics make this ion very useful in illuminations, TV color screen, and displays to diode productions.

The recent interest in study the organic-inorganic hybrid materials is motivated by the presence of their extraordinary properties which

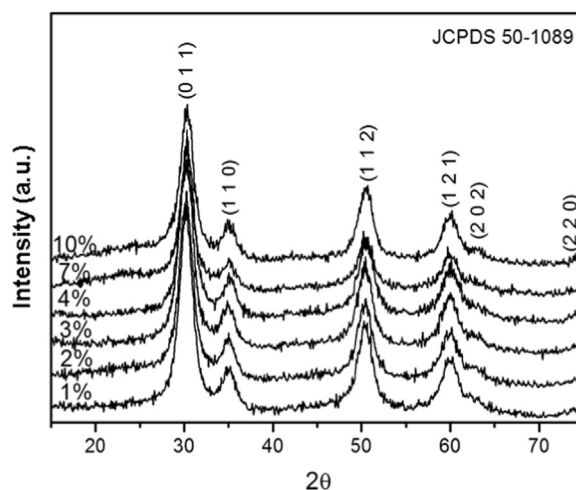


Fig. 3. Powder X-ray diffractograms of the  $ZrO_2:Hhpa:Eu^{3+}$  presenting different  $Eu^{3+}$  concentration: 1.0%, 2.0%, 3.0%, 4.0%, 7.0%, and 10.0%.

promote their applicability in different field, since this material combine the mutual vantages from the organic and inorganic material [1,24]. Hybrid materials formed with  $Eu^{3+}$  and one organic ligand are used to fabricate red luminophors for the obtention of white light emitter diodes (LEDs), since the emission of the white light can be obtained by the incorporation of blue, green, and red luminophors. The organic ligands act as antenna, which increase the luminescent intensity due to the presence of chromophores groups [1,25–29].

The organic compound 3-hydroxypicolinamide (Hhpa), Fig. 1, is an aromatic ligand considered as a model compound that is a component of the virginiamycine S (VS), an hexapeptide which act as an antibiotic that promotes the blocking of the protein synthesis into the bacterial ribosome. The 3-hydroxypicolinil residue absorbs the light and is also responsible for the complexant properties and to the phenomenon of the transference of the protons to the peptide VS. It is believed that the complexation of the species hpa (deprotonated ligand) with the rare

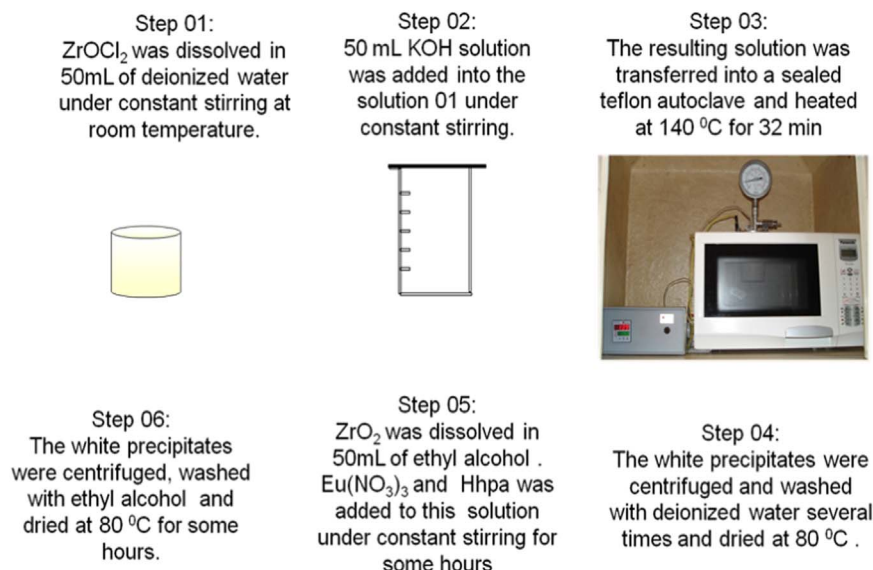


Fig. 2. Experimental procedure for preparation of the samples  $ZrO_2$ ,  $ZrO_2:Hhpa:Eu$ , as well  $ZrO_2:Bipy:Eu$ .

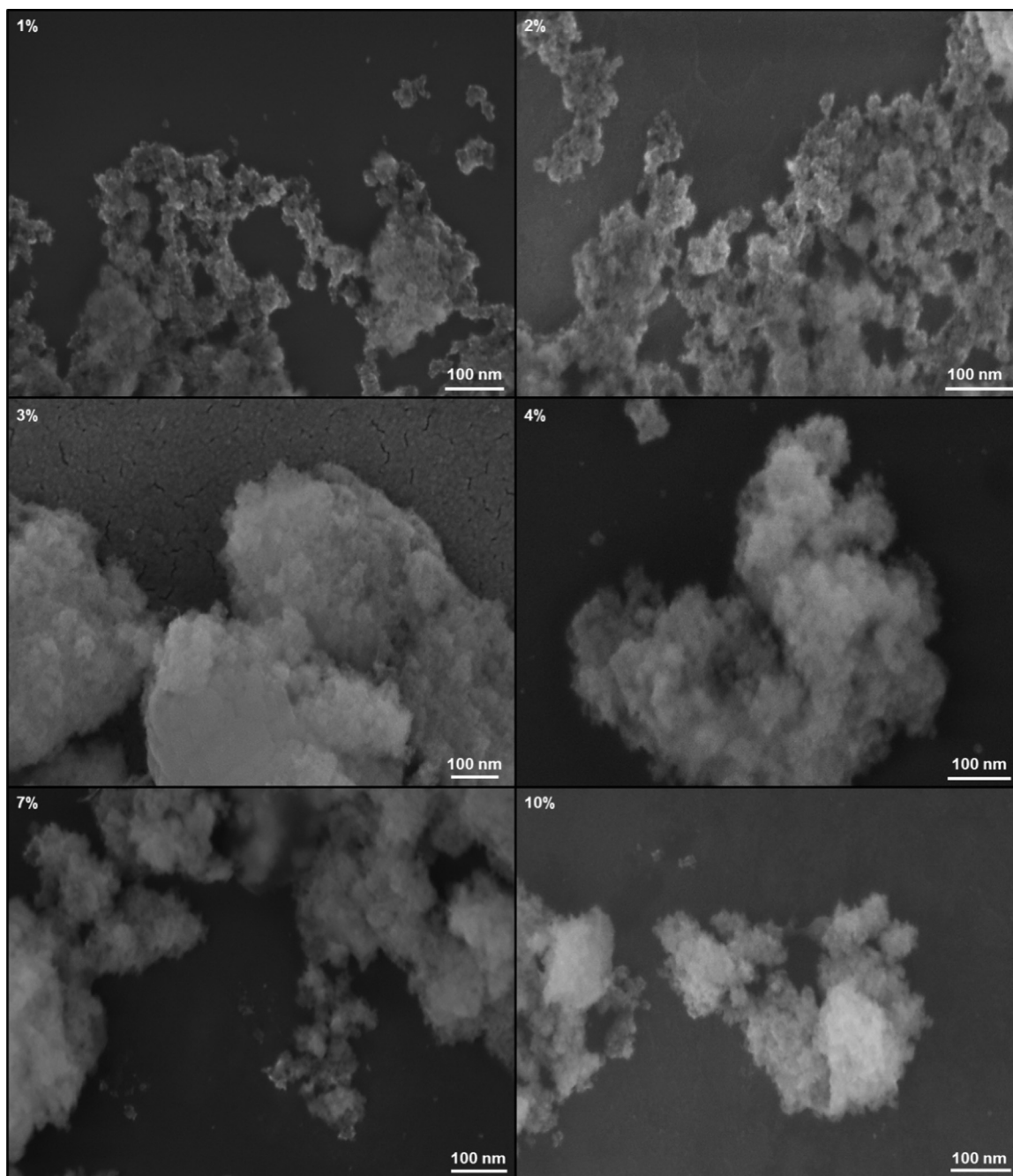


Fig. 4. FE-SEM microographies of  $\text{ZrO}_2\text{:Hhpa:Eu}^{3+}$  presenting different  $\text{Eu}^{3+}$  concentration (a) 1.0%, (b) 2.0%, (c) 3.0%, (d) 4.0%, (e) 7.0%, and (f) 10.0%.

alkaline ions occurs through the oxygen from the phenolate group with the participation of the amide from the carboxylic group [11].

In this work  $\text{ZrO}_2$  was prepared by the hydrothermal microwave method (HMM) at  $140^\circ\text{C}$  for 32 min. The hybrid materials,  $\text{ZrO}_2\text{:Hhpa:Eu}^{3+}$ , were obtained using the previous  $\text{ZrO}_2$  dispersed in distilled water, in which was added 1.0%, 2.0%, 3.0%, 4.0%, 7.0%, and 10.0% in mass of  $\text{Eu}^{3+}$ , as well as the ligand 3-hydroxypicolinamide (Hhpa). These solutions were stirring for about 3 h, being filtered and washed several times with distilled water and ethanol, and then dried overnight at  $60^\circ\text{C}$ . These samples were analyzed by powder X-ray diffraction (XRD), Scanning Electron Microscopy (SEM) and Transmission

electron microscopy (TEM). Their photoluminescent properties were investigated through their excitation and emission spectra. The behavior of these materials was also performed for gas sensors.

## 2. Experimental section

### 2.1. Preparation of hybrid material

The samples were prepared according to the steps represented in Fig. 2.

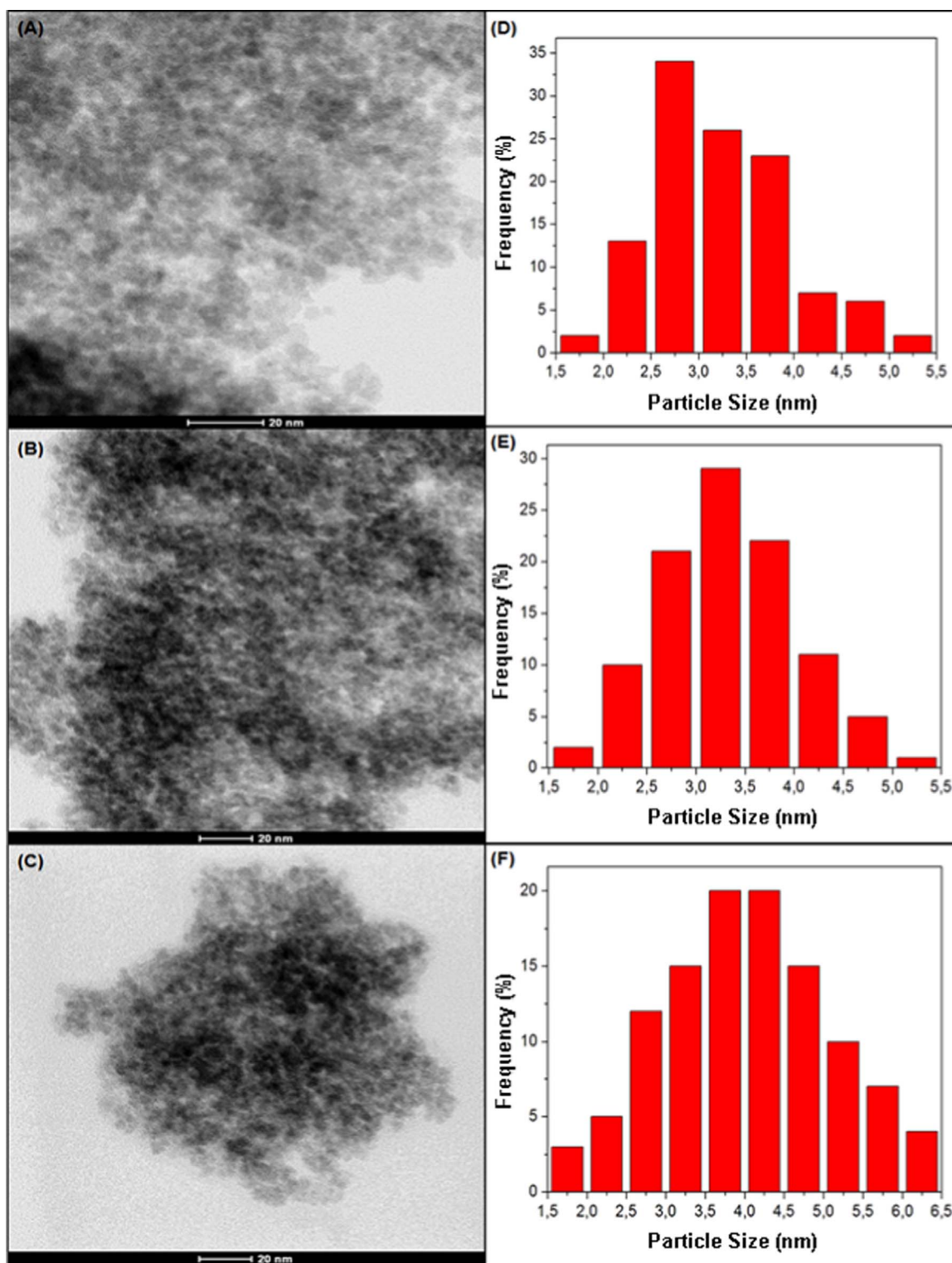


Fig. 5. TEM micrographies and the distribution of the average sizes of the particles for the sample  $\text{ZrO}_2\text{:Hhpa:Eu}^{3+}$  with (A) 1.0%, (B) 2.0%, and (C) 3.0% of  $\text{Eu}^{3+}$ .

## 2.2. Structural characterization

Powder X-ray diffraction (XRD) measurements were performed on a Rigaku-D/max 2500 diffractometer (Japan) with graphite-monochromatized  $\text{Cu K}\alpha$  radiation ( $\lambda = 0.15405 \text{ nm}$ ). Scan was made from  $15^\circ$  to  $75^\circ$  with a scanning velocity of  $2^\circ/\text{min}$ . PL data of the Eu-doped powders were obtained in a Jobin Yvon – Fluorolog III spectrofluorometer under excitation of a 450 W xenon lamp. Luminescence

lifetime measurements were carried out as well using a 1940D model spectrophosphorometer coupled to the spectrofluorometer. All the PL measurements were performed at room temperature. The shape and size of the crystals were observed by Field Emission Scanning Electron Microscopy (FE-SEM) through a Carl Zeiss microscope (Model Supra 35) operated at an accelerating voltage of 30 kV and a working distance of 3.7 mm. TEM, at 200 kV measurements were performed using an FEI microscope (model Tecnai G2 F20, FEI, Hillsboro, OR).

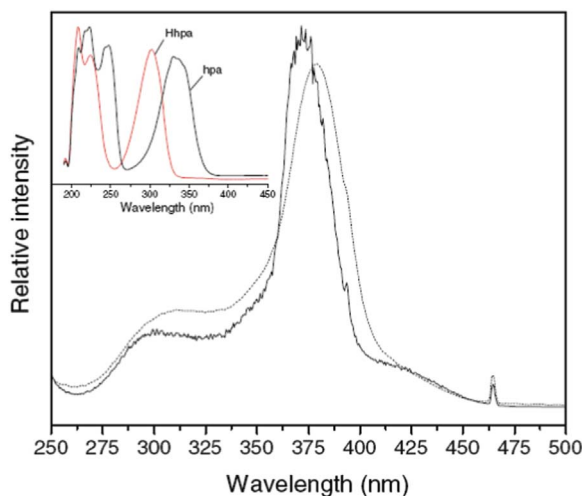


Fig. 6. Excitation spectra of  $\text{ZrO}_2\text{:Hhpa:Eu}^{3+}$  ( $\lambda_{\text{Em.}} = 612$  nm) (solid line) at 77 K and at room temperature (dashed line). (Inset: absorption spectra of the ligand Hhpa and hpa species at ethanolic solution).

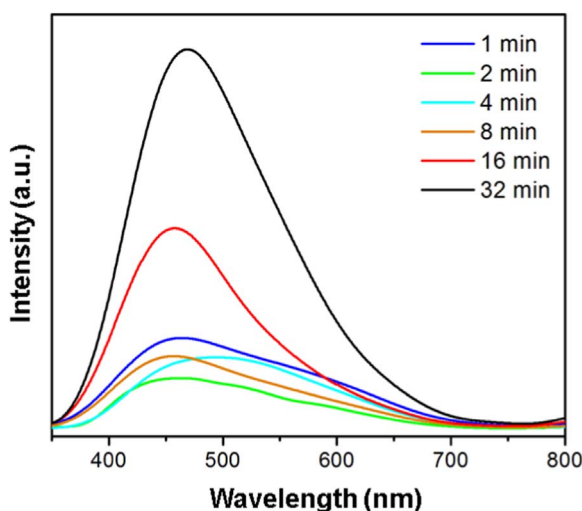


Fig. 7. Emission spectra ( $\lambda_{\text{Exc.}} = 375$  nm, room temperature) of the  $\text{ZrO}_2$  powders obtained under microwave heat treatment at 140 °C for 1, 2, 4, 8, 16 and 32 min.

### 3. Results and discussion

The Powder X-ray diffractograms of the  $\text{ZrO}_2\text{:Hhpa:Eu}^{3+}$  presenting different  $\text{Eu}^{3+}$  concentration: 1.0%, 2.0%, 3%, 4.0%, 7.0%, and 10.0%. XRD of  $\text{ZrO}_2$  and  $\text{ZrO}_2\text{:Hhpa:Eu}^{3+}$  are showed at Fig. 3.

The diffractograms presented at Fig. 3 show the same behavior for the species  $\text{ZrO}_2\text{:Eu}^{3+}$ , as well as for the  $\text{ZrO}_2\text{:Hhpa:Eu}^{3+}$  samples presenting different  $\text{Eu}^{3+}$  concentration: 1.0%, 2.0%, 3.0%, 4.0%, 7.0%, and 10.0%. The presence of  $\text{Eu}^{3+}$  complexes did not changed the crystallographic behavior of pure  $\text{ZrO}_2$  powder. It was noticed that all of materials show the tetragonal phase, and the diffractograms possess the peaks related to this phase at  $30.2^\circ$  (0 1 1),  $34.8^\circ$  (1 1 0),  $50.2^\circ$  (1 1 2),  $59.8^\circ$  (1 2 1),  $62.7^\circ$  (2 0 2),  $74.1^\circ$  (2 2 0), according to the JCPDS (Joint Committee on Powder Diffraction Standards) no 50-1089 from the space group  $P42/nmc$  [5].

Fig. 4 presents the FE-SEM micrographies of  $\text{ZrO}_2\text{:Hhpa:Eu}^{3+}$  presenting different  $\text{Eu}^{3+}$  concentrations. This technique was used to observe the morphology of the materials and to identify possible morphological alterations when different organic ligands are employed.

According to these images we can conclude the particles are

agglomerated for all of the samples. The particles are nanosized and the presence of the organic ligands did not change significantly the morphology of  $\text{ZrO}_2$  powders.

The particle sizes for the samples were evaluated through the TEM analysis, since this technique presents a better resolution than SEM. The TEM micrographies and the distribution of the average sizes of the particles for the sample  $\text{ZrO}_2\text{:Hhpa:Eu}^{3+}$  with 1.0% – (A) and (D), 2.0% – (B) and (E), and 3.0% – (C) and (F) of  $\text{Eu}^{3+}$  are presented in Fig. 5, respectively. These results show the average particle sizes for the samples of around 2.5–4.5 nm. The difference in  $\text{Eu}^{3+}$  concentration did not alter significantly these values.

Fig. 6 presents the excitation spectra of  $\text{ZrO}_2\text{:Hhpa:Eu}^{3+}$  ( $\lambda_{\text{Em.}} = 612$  nm) (solid line) at 77 K and at room temperature (dashed line). Inset: absorption spectra of the ligand Hhpa and hpa species at ethanolic solution. It is observed a broad excitation band at around 375 nm when the sample was monitored setting the emission at the wavelength of 612 nm. Similar feature was observed for both temperatures used during the acquisition of these excitation spectra [1].

The emission spectra ( $\lambda_{\text{Exc.}} = 375$  nm, at room temperature) of the  $\text{ZrO}_2$  powders obtained under microwave heat treatment at 140 °C for 1, 2, 4, 8, 16 and 32 min are presented in Fig. 7. These spectra present broad bands covering a large area in the visible range of the electromagnetic spectra indicating that in the photoluminescence emission process are involved different intermediate energy levels in a multiphonon mechanism. The maxima wavelengths for the photoluminescent emission of the samples are centered at around 480 nm, corresponding to the blue emission.

Fig. 8(A) shows the excitation spectra of the  $\text{ZrO}_2\text{:Hhpa:Eu}^{3+}$  (a) 1.0%, (b) 2.0%, (c) 3.0%, (d) 4.0%, (e) 7.0%, and (f) 10.0%; while Fig. 8(B) presents in 2D way the emission spectra of the sample  $\text{ZrO}_2\text{:Hhpa:Eu}^{3+}$ , where the  $\text{Eu}^{3+}$  concentrations are (a) 1.0%, (b) 2.0%, (c) 3.0%, (d) 4.0%, (e) 7.0%, and (f) 10.0% in weight, using both excitation wavelength of 362 and 393 nm.

It is possible to notice at Fig. 8(A) the  $\text{Eu}^{3+}$  characteristic excitation bands which maxima are observed at 360, 384, 393 and 464 nm, related to the  $\text{Eu}^{3+}$   ${}^7\text{F}_0 \rightarrow {}^5\text{D}_4$ ,  ${}^7\text{F}_0 \rightarrow {}^5\text{G}_1$ ,  ${}^7\text{F}_0 \rightarrow {}^5\text{L}_6$  and  ${}^7\text{F}_0 \rightarrow {}^5\text{D}_2$  transitions, respectively. The large band at around 360 nm are ascribed to the energy transfer from the Hhpa ligand to the  $\text{Eu}^{3+}$  in the  $\text{ZrO}_2\text{:Hhpa:Eu}^{3+}$  hybrid materials. The antenna effect when the organic ligand is present in  $\text{ZrO}_2$  powders is very effective.

The data presented at Fig. 8(B) show the characteristic emission bands of the  $\text{Eu}^{3+}$  transitions, which maxima intensities are observed at 580 nm ( ${}^5\text{D}_0 \rightarrow {}^7\text{F}_0$ ), 593 nm ( ${}^5\text{D}_0 \rightarrow {}^7\text{F}_1$ ), 613 nm ( ${}^5\text{D}_0 \rightarrow {}^7\text{F}_2$ ), 652 nm ( ${}^5\text{D}_0 \rightarrow {}^7\text{F}_3$ ), and 702 nm ( ${}^5\text{D}_0 \rightarrow {}^7\text{F}_4$ ) for all of the samples. The samples excited at 362 nm showed more intensity transitions in its emission spectra compared to the samples excited at 394 nm, confirming an efficient energy transfer from the ligand to the  $\text{Eu}^{3+}$  in the organic-inorganic hybrid materials.

The decay curves of the  $\text{Eu}^{3+}$   ${}^5\text{D}_0 \rightarrow {}^7\text{F}_2$  transition of the samples  $\text{ZrO}_2\text{:Hhpa:Eu}^{3+}$  are presented at Fig. 9 ( $\lambda_{\text{em.}} = 615$  and  $\lambda_{\text{exc.}} = 393$  nm). The decay curves for all of the samples present biexponential behavior and can be adjusted as the following equation:

$$I = I_1 e^{-(t/T_1)} + I_2 e^{-(t/T_2)}$$

where,  $I_1$  and  $I_2$  are the intensities for both different decay times ( $t$ ),  $T_1$  and  $T_2$ , respectively. This indicates that the  $\text{Eu}^{3+}$  ions are localized in at least two different  $\text{Eu}^{3+}$  sites. As expected for the f-f transitions the lifetime values for these materials were in the microseconds scale, which values are showed in Table 1 [2].

According to the data presented at Figs. 8 and 9, and Table 1, the obtention of the hybrid materials originated from the  $\text{Eu}^{3+}$  and the ligand 3-hydroxipicolinamide (Hhpa) in the nanometric crystals of the  $\text{ZrO}_2$  powders was proved. The study of the photoluminescent properties of these materials shows the  $\text{Eu}^{3+}$  complex was formed. When the samples were excited at 363 nm the ligand absorbs efficiently the light

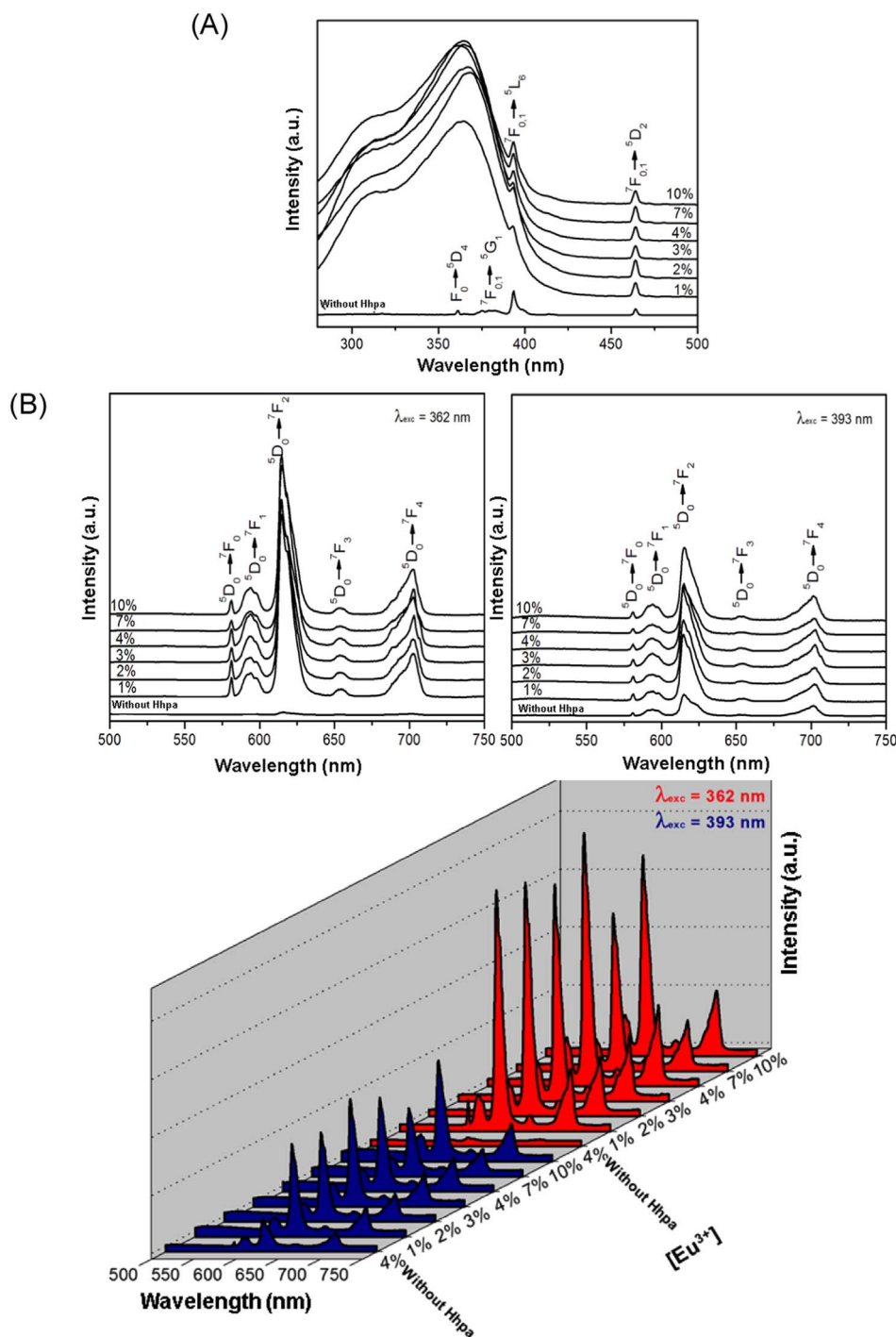


Fig. 8. (A) Excitation spectra of the complex Bipy:Eu<sup>3+</sup>, ( $\lambda_{Em} = 615$  nm) at room temperature and ZrO<sub>2</sub>:Hhpa:Eu<sup>3+</sup> presenting different Eu<sup>3+</sup> concentrations: (a) 1.0%, (b) 2.0%, (c) 3%, (d) 4.0%, (e) 7.0%, and (f) 10.0%. (B) Emission spectra of the samples ZrO<sub>2</sub>:Eu<sup>3+</sup> and ZrO<sub>2</sub>:Hhpa:Eu<sup>3+</sup>, where the Eu<sup>3+</sup> concentrations are 1.0%, (b) 2.0%, (c) 3.0%, (d) 4.0%, (e) 7.0%, and (f) 10.0% in weight, using both excitation wavelengths,  $\lambda_{exc}$ , of 362 and 393 nm.

at this wavelength and transfer part of this energy to the Eu<sup>3+</sup> that emit their characteristic red emission at 613 nm in a higher intensity when compared to the samples without this organic ligand. The lifetime values of the ZrO<sub>2</sub>:Hhpa:Eu<sup>3+</sup> with different weight % of Eu<sup>3+</sup> are bi-exponential fitted (Table 1). The sample without the organic ligand presents a lifetime of 0.07 ms while the sample ZrO<sub>2</sub>:Hhpa:Eu<sup>3+</sup> with 2.0% of Eu<sup>3+</sup> presented a lifetime value of around seven times bigger (0.46 ms) than the sample without the organic ligand. Another finding was that the nanoparticles of ZrO<sub>2</sub> were seen in a spherical pattern and

agglomerated through the FE-SEM (Fig. 4) and TEM analysis (Fig. 5). The particles presented sizes of around 3 nm and the difference in the Eu<sup>3+</sup> concentration did not affect in a significant way the structure, morphology neither the optic properties of these materials.

The measurement of the sample colors were evaluated by the Commission internationale de l'éclairage (CIE) coordinates. The CIE diagram of the ZrO<sub>2</sub> samples under different synthesis conditions are plotted in the CIE chromaticity diagram based on the photoluminescence emission spectra. The CIE diagram is shown in Fig. 10(a –

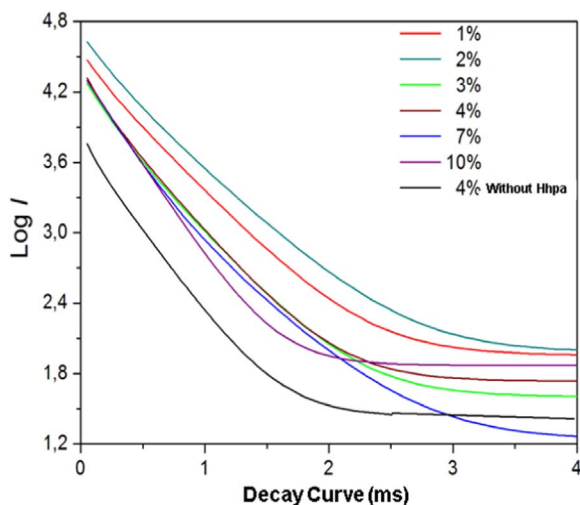


Fig. 9. Intensity versus time (ms) of the  $^5D_0 \rightarrow ^7F_2$  transition of the samples  $ZrO_2:Hhpa:Eu^{3+}$  with different % of  $Eu^{3+}$ .

Table 1  
Lifetime  $T_1$  and  $T_2$  (in ms) of the powders  $ZrO_2:Eu^{3+}$  and  $ZrO_2:Hhpa:Eu^{3+}$  with different weight % of  $Eu^{3+}$ .

% $Eu^{3+}$	$\tau_1$ (ms)	$\tau_2$ (ms)
1.0%	0.14	0.40
2.0%	0.46	0.19
3.0%	0.17	0.39
4.0%	0.36	0.12
7.0%	0.23	0.55
10.0%	0.22	0.48
4% without Hhpa	0.07	0.29

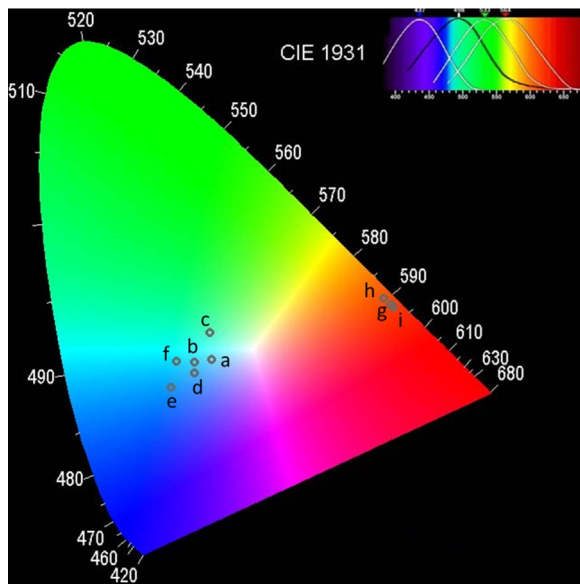


Fig. 10. CIE diagram of the  $ZrO_2$  samples under different synthesis conditions. (a)  $ZrO_2$  1 min, (b)  $ZrO_2$  2 min, (c)  $ZrO_2$  4 min, (d)  $ZrO_2$  8 min, (e)  $ZrO_2$  16 min, (f)  $ZrO_2$  32 min, (g)  $ZrO_2:Hhpa:3\%Eu^{3+}$ ,  $\lambda_{exc} = 362$  nm, (h)  $ZrO_2:Hhpa:3\%Eu^{3+}$ ,  $\lambda_{exc} = 393$  nm, and (i)  $ZrO_2:3\%Eu$  without Hhpa,  $\lambda_{exc} = 393$  nm.

i), and the respective CIE chromatic coordinates of these samples are listed in Table 2. The  $ZrO_2$  samples synthesized in different time shows predominant color in the blue-green region of the diagram. The sample  $ZrO_2$  heated for 4 min has the emission in the green region while the

Table 2  
CIE chromatic coordinates of the  $ZrO_2$  samples under different synthesis conditions Lifetime  $T_1$  and  $T_2$  (in ms) of the powders  $ZrO_2:Eu^{3+}$  and  $ZrO_2:Hhpa:Eu^{3+}$  with 3.0 wt% of  $Eu^{3+}$ .

Samples	Chromatic coordinates	
	x	y
(a) $ZrO_2$ 1 min	0.282	0.318
(b) $ZrO_2$ 2 min	0.254	0.314
(c) $ZrO_2$ 4 min	0.279	0.361
(d) $ZrO_2$ 8 min	0.255	0.297
(e) $ZrO_2$ 16 min	0.217	0.273
(f) $ZrO_2$ 32 min	0.226	0.316
(g) $ZrO_2:3\%Eu$ $\lambda_{exc} = 362$ nm	0.571	0.407
(h) $ZrO_2:3\%Eu$ $\lambda_{exc} = 393$ nm	0.559	0.417
(i) $ZrO_2:4\%Eu$ without Hhpa $\lambda_{exc} = 393$ nm	0.575	0.403

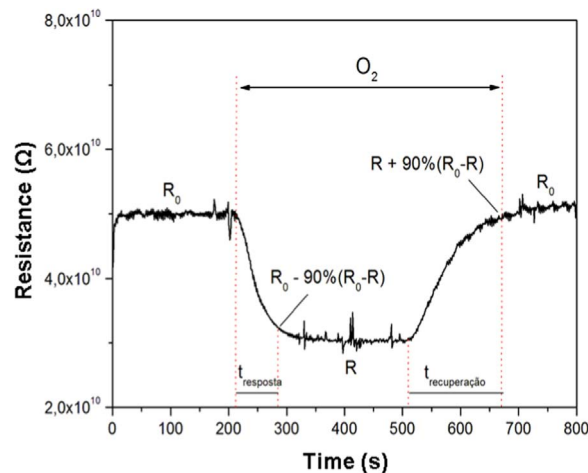


Fig. 11. Illustration of the procedure used to determine the response and recovery time for the representative sample of the material  $ZrO_2:Hhpa:Eu^{3+}$  exposed to 50 ppm of  $O_2$  for an operating temperature ( $T_{opt}$ ) of 300 °C.

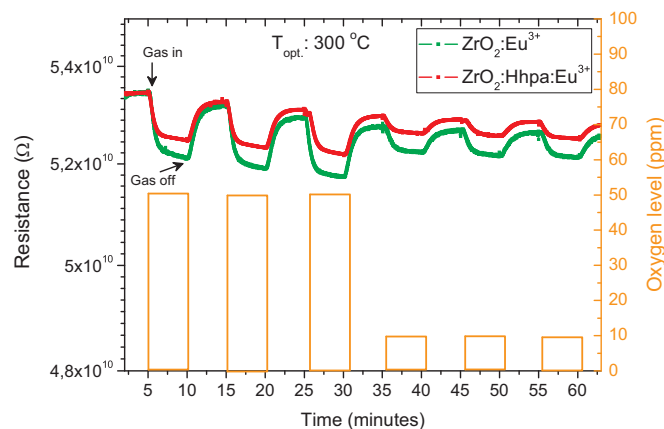


Fig. 12. Electrical resistance of  $ZrO_2:Eu^{3+}$  and  $ZrO_2:Hhpa:Eu^{3+}$  samples upon exposure to 50 and 10 ppm of  $O_2$ , at an operating temperature of 300 °C.

emission of the sample  $ZrO_2$  heated for 16 min is more close to the blue color. The samples  $ZrO_2:Hhpa:Eu^{3+}$  doped with  $Eu^{3+}$ , on the other hand, presented intense orange color. Both excitation wavelengths (362 and 393 nm) were sufficient to sensitize the  $Eu^{3+}$  transitions responsible for their red color. The main intense red color arises specially from the  $^5D_0 \rightarrow ^7F_2$   $Eu^{3+}$  transition observed at 613 nm in the emission spectra. Therefore, the contribution of the blue color to the host lattice

**Table 3**

Values of response and recovery time for the  $\text{ZrO}_2:\text{Eu}^{3+}$  at 300 °C and  $\text{ZrO}_2:\text{Hhpa}:\text{Eu}^{3+}$ , synthesized with 3,0 wt% of  $\text{Eu}^{3+}$  at 200 and 300 °C, under flux of 10 and 50 ppm of  $\text{O}_2$ .

Samples	Operating temperature, $T_{\text{opt}}$ (°C)	Response time, $t_{\text{res}}$ (s)		Recovery time, $t_{\text{rec}}$ (s)	
		10 ppm	50 ppm	10 ppm	50 ppm
$\text{ZrO}_2:\text{Eu}^{3+}$	300	108	102	141	124
$\text{ZrO}_2:\text{Hhpa}:\text{Eu}^{3+}$	300	126	106	132	129
$\text{ZrO}_2:\text{Hhpa}:\text{Eu}^{3+}$	200	93	71	140	141

and the red color of the  $\text{Eu}^{3+}$  are responsible for the dark orange color of these samples. The values presented at Table 2 shows that the sample  $\text{ZrO}_2:3\%\text{Eu}$  without Hhpa,  $\lambda_{\text{exc.}} = 393$  nm, presented the values ( $x = 0.575$ ;  $y = 0.403$ ), which are very close to the commercial phosphor  $\text{Y}_2\text{O}_3:\text{S}:\text{Eu}^{3+}$  values ( $x = 0.64$ ;  $y = 0.34$ ), published for the International Standards Chromaticity Coordinates [30]. These results confirm these materials as suitable for applications in several visible lamps, displays and other optical devices.

The study of these materials for application as gas sensors [31–33] was evaluated through the variation of the electric resistance of the nanostructures of  $\text{ZrO}_2$ ,  $\text{ZrO}_2:\text{Eu}^{3+}$  and  $\text{ZrO}_2:\text{Hhpa}:\text{Eu}^{3+}$  containing 3% of  $\text{Eu}^{3+}$  in the presence of different oxygen concentrations (10 and 50 ppm) at 200 and 300 °C. The sensor response ( $S$ ) was defined as  $S = R_{\text{air}} / R_{\text{oxygen}}$ , where  $R_{\text{air}}$  and  $R_{\text{oxygen}}$  are the electrical resistances of the sensor upon exposure to dry air and oxygen gas, respectively. The sensor response time ( $\tau_{\text{res}}$ ) was defined as the time required for the electrical resistance to reach 90% of the initial value when exposed to oxygen gas. In the same way, the recovery time ( $\tau_{\text{rec}}$ ) was defined as the time required for the electrical resistance to recover 90% of the initial value after the oxygen gas flow was switched off.

The velocity of the sensor response was measured through the response time and recovery time. Fig. 11 shows the illustration of the procedure used to determine the response and recovery time for the  $\text{ZrO}_2:\text{Hhpa}:\text{Eu}^{3+}$  at 300 °C under  $\text{O}_2$  flux of 50 ppm.

The dynamic sensor resistance of  $\text{ZrO}_2:\text{Hhpa}:\text{Eu}^{3+}$  samples exposed to oxygen gas is illustrated in Fig. 11. Upon exposure to oxygen gas, the electrical resistance of samples quickly decreased, which is a typical behavior of p-type semiconductors exposed to oxidizing gases [34]. Additionally, the 3-hydroxypicolinamide (Hhpa) addition into the  $\text{ZrO}_2:\text{Eu}^{3+}$  network did not alter the conduction mechanism of the sample, as seen in Fig. 12. The reversible cycles of the sensor curves indicate a stable and reproducible response for both samples. In addition, the sensor response was slightly reduced with the Hhpa addition into the  $\text{ZrO}_2:\text{Eu}^{3+}$ , as seen in Fig. 12. The response and recovery times are displayed in Table 3.

The analysis of Figs. 12 and 13 reveals a significant reduction of sensor response with operating temperature from  $S = 1.02$  (300 °C) to

$S = 1.62$  (200 °C), as displayed in Table 3. For temperature below 200 °C, the sample was not sensitivity for this oxygen concentration range. Additionally, even for a lower temperature, the  $\text{ZrO}_2:\text{Hhpa}:\text{Eu}^{3+}$  sample exhibit a good reproducibility, and a total reversibility.

The results above-discussed show clearly that the hybrid  $\text{ZrO}_2:\text{Hhpa}:\text{Eu}^{3+}$  compound can be considered as a potential oxygen gas sensor. Nevertheless, the work has demonstrated the multifunctional properties of  $\text{ZrO}_2:\text{Hhpa}:\text{Eu}^{3+}$  compound. Further gas-sensing parameters still need to be evaluated, such as, its selectivity to reducing gases, as well as its long-term stability.

#### 4. Conclusions

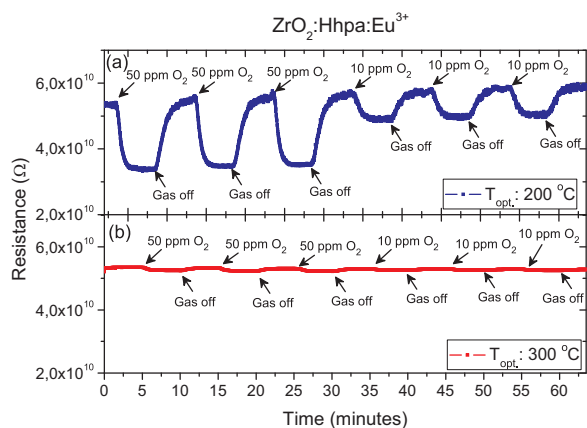
It was obtained for the first time the completely crystalline powders where XRD peaks were perfectly indexed to the tetragonal phase of  $\text{ZrO}_2$  (JCPDS 50-1089). TEM micrographs revealed a spherical nanostructure pattern presenting an average diameter of 4 nm. Excitation and emission spectra revealed the characteristic peaks of the  $\text{Eu}^{3+}$  transitions. For the first time it is showed here that the ligand plays an important role in transfer energy to  $\text{Eu}^{3+}$  ion, known as antenna effect, increasing the  $\text{Eu}^{3+}$  emission intensity of the samples  $\text{ZrO}_2:\text{Hhpa}:\text{Eu}^{3+}$  when compared to the species  $\text{ZrO}_2:\text{Eu}^{3+}$ . The studies of this materials as gas sensors showed the hybrid  $\text{ZrO}_2:\text{Hhpa}:\text{Eu}^{3+}$  compound can also be considered as a potential oxygen gas sensor.

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**Fig. 13.** Electrical resistance of  $\text{ZrO}_2:\text{Hhpa}:\text{Eu}^{3+}$  sample upon exposure to 50 and 10 ppm of  $\text{O}_2$ , at (a)  $T_{\text{opt.}}$  of 200 °C, and (b)  $T_{\text{opt.}}$  of 300 °C.



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