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## Surface Modification of the Alloy Ti-7.5Mo by Anodization for Biomedical Applications

Ana Lucia do Amaral Escada<sup>1,a</sup>, Javier Andres Munoz Chaves<sup>1,b</sup>,  
Ana Paula Rosifini Alves Claro<sup>1,c</sup>

<sup>1</sup> Univ. Estadual Paulista - UNESP, Department of Materials and Technology,  
Faculty of Engineering Guaratinguetá, Av. Dr. Ariberto Pereira da Cunha, 333, Pedregulho,  
CEP 12.516-410, Guaratinguetá, SP, Brazil.

<sup>a</sup> analuciaescada@uol.com.br, <sup>b</sup> javier\_munozch@outlook.com, <sup>c</sup> rosifini@feg.unesp.br

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**Abstract.** The purpose of this study was to evaluate the TiO<sub>2</sub> nanotubes growth and the variation in its diameter to improve the surface properties of Ti-7.5Mo to use for biomedical applications. For the nanotubes TiO<sub>2</sub> growth, the samples were anodized in glycerol and ammonium fluoride and divided according to the anodizing potential at 5V to 10V and 24 hour time. The surfaces were examined by scanning electron microscope (SEM), X-ray analysis (XRD) and contact angle measurements. The average tube diameter, ranging in size from 13 to 23 nm, was found to increase with increasing anodizing voltage. It was also observed a decrease in contact angle in accordance with the increase in the anodizing potential. The X-ray analysis showed the presence of anatase phase in samples whose potential was 10V and this condition represents a simple surface treatment for Ti-7.5Mo alloy that has high potential for biomedical applications.

### Introduction

Titanium (Ti) and its alloys are widely used in dental and orthopedic applications due light weight and appropriate mechanical properties. Also, titanium-based alloys have excellent corrosion resistance and good biocompatibility [1].

The surface properties of materials are a very important factor to occur a good osseointegration and the topography and surface physicochemistry are elements that enhance the development of the bone around an implant [2]. To occur a better biomedical implant osseointegration is necessary to have a surface that promotes biological and cellular processes.

Over the past decade, various techniques such as micromachining [3], grinding, polishing [4], and chemical methods like acid etching [5], alkali etching [6], and anodization [7] of titanium surface modification have been employed to fabricate implant surfaces. In recent years, a nanometer implants surface has been paramount to the survival of cells and tissue acceptance [8].

Anodization is an electrolytic passivation technique used to increase the thickness of the natural oxide layer on metal surfaces. This technique has attracted great attention in recent years due to its simplicity as well as reproducibility of the results obtained [9,10]. The thickness and structure of the oxide layer formed (amorphous or crystalline) depends on the applied potential between the electrodes and duration of anodization time. The structure of the oxide film formed on titanium can be anatase, a mixture of anatase and rutile, or rutile [11].

Recent studies have demonstrated that controlled anodizing titanium alloys leads to the formation of a layer of self-assembled nanotubes [12]. The diameter of these nanotubes can be controlled by anodization potential as demonstrated by Bauer et al. 2006. In their work, different anodizing potentials (from 1V and 25V) were applied and self-organized porous structures with a diameter from 15 nm to 120 nm and length from 20 nm to 1 um were formed [13]. In another study, Park et al. 2007 demonstrated that the response of mesenchymal cells were dependent on the diameter of the TiO<sub>2</sub> nanotubes. They obtained nanotubes with diameter between 15 and 100 nm be able to accelerate cell activity when compared to other diameters and a smooth surface. [14].

In the present study the variation in the diameter of nanotube TiO<sub>2</sub> was investigated according to the applied anodizing potential order to obtain TiO<sub>2</sub> nanotubes with diameters that improve the surface properties of Ti-7.5Mo to use for biomedical applications. This alloy was chosen due to their excellent bulk properties, a low elastic modulus and a high strength/modulus ratio in order to obtain a better surface.

### Materials and methods

The Ti-7.5Mo alloy was produced from sheets of commercially pure titanium (99.9%) and molybdenum (99.9%). Samples were melted in an arc furnace under an argon atmosphere. Then, the ingots were then homogenized under vacuum at 1100°C for 86.4 ks to eliminate chemical segregation. They were cold worked by swaging and bars with 13 mm of diameter were produced. Then discs with 4 mm of thickness were cut and samples were divided into two groups according anodization potential.

Samples were grided with emery papers (200 to 1200) and polished with a solution formed by colloidal silica (OPS – Struers) plus 5% oxalic acid. They were clean in the ultrasonic bath. Anodization process was performed at 5 V and 10 V for 24 hours. After the anodization, the samples were washed with deionized water, dried, and calcined at 450 °C at a heating rate of 5 °C per minute, 1 hour of permanence. And cooled in the oven.

Surfaces were evaluated using a scanning electron microscope (SEM-FEG, XL 30 FEG, Philips) after anodizing.

X-ray diffraction (XRD) for phase analysis were realized using a Shimadzu diffractometer (Shimadzu-XRD 6000) operated using Cu K $\alpha$  radiation (40kV / 30mA) with scan mode continuous and scan rate of 1.0 deg/min

The wettability was evaluated by contact angle measurements. The contact angle was obtained by using the sessile drop method on an advanced Rame-Hart goniometer, model n° 300-F1. A microliter syringe pump was attached to a small needle in an XYZ manipulator to enable a drop to be slowly increased and decreased in size. The shape of the drop was recorded by a digital camera and the contact angles were measured from the images. The volume of each drop was 2  $\mu$ l and the average value of at least 5 drops was calculated.

### Results and discussion

Figure 1 shows the micrographs of the samples after anodization at 5 V and 10 V for 24 hours. All samples were calcined at 450 °C. A self-organized layer and homogeneous of the nanotubes was obtained for all conditions evaluated. For samples anodized at 5 V (Fig 1a) and 10 V (Fig 1b), the average pore diameter was 13.68 nm and 22.83 nm, respectively. It could be concluded that the increase of the voltage and diameter are related.

These results were similar with Lockman *et al.* (2010) [15] who concluded that the tube diameter is linearly dependent on the applied potential during the growth of nanotubes. According Macak *et al.* (2008) [16] either, diameter changes linearly with the applied potential during the growth of nanotubes and it was also observed in our results. Furthermore, the diameter of the nanotube in the condition of 5V is smaller (13.68 nm) than for 10V (22.83 nm) and the nearest ideal condition for adhesion and proliferation cell, according to Park *et al.* 2007 [14].

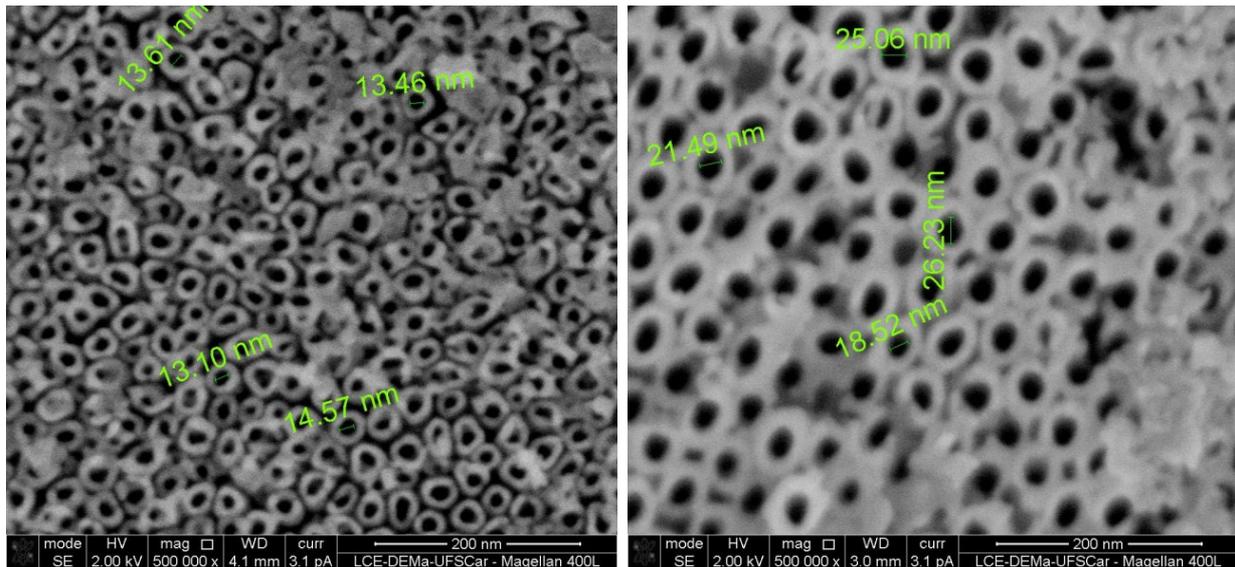


Fig. 1 - Field emission gun scanning electron microscopy images (SEM-FEG): (a) anodized Ti-7.5Mo - 5V-24h-450°C; (b) anodized Ti-7.5Mo - 10V-24h-450°C

Figure 2 shows the X-ray diffraction patterns for samples anodized at 5 V and 10 V for 24 hours and calcined at 450 °C. Structural analyses shows that the surface consisted of a TiO<sub>2</sub> layer in amorphous state and diffraction peaks related to the substrate ( $\alpha$ -Ti phase) in the XRD pattern were observed for both anodization (5 V and 10 V), but the X-ray analysis showed the presence of anatase phase (25°, 48°, 53° and 54°) is only present for samples whose potential is 10V. The obtained peaks in XRD pattern are consistent with JCPDS Card No. 84-1286.

According Park et al. 2007 [14], TiO<sub>2</sub> exists in three different allotropic form, anatase, rutile and brookite, therefore the TiO<sub>2</sub> nanotubes can also be found in these different phases, but the anatase most desired to be biocompatible. In this study, the anatase phase is present only in samples anodized at 10V, and its appearance is probably due to the greater diameter of the TiO<sub>2</sub> nanotubes formed during the application of this potential, making this to the better condition for biomedical applications.

The graphic 1 showed the contact angle measurements for two conditions studied. For samples anodized at 5 V, the contact angle was 7.8°, and for samples anodized at 10 V, it was 15.3°. It is possible to observe that in this case, the increasing of the anodization potential increased the nanotube diameter and the contact angle. The samples anodized at 5V showed that the nanotube diameter was 13.68 nm with contact angle of 7.8°, while the samples anodized at 10V showed that the nanotube diameter was 22.83 nm with contact angle of 15.3°. Surface wettability (hydrophobicity/hydrophilicity) is one of the most important parameters affecting the biological response of an implanted biomaterial. Wettability affects protein adsorption, platelet adhesion/activation, blood coagulation and cell and bacterial adhesion [17-18]. Highly hydrophilic surfaces seem more desirable than hydrophobic ones in view of their interactions with biological fluids, cells and tissues [19]. According to Lim and Donahue [20], the relationship between the contact angle and wetting occurs in reverse on the same surface. Therefore, a decrease of this angle increases the capacity for surface wettability.

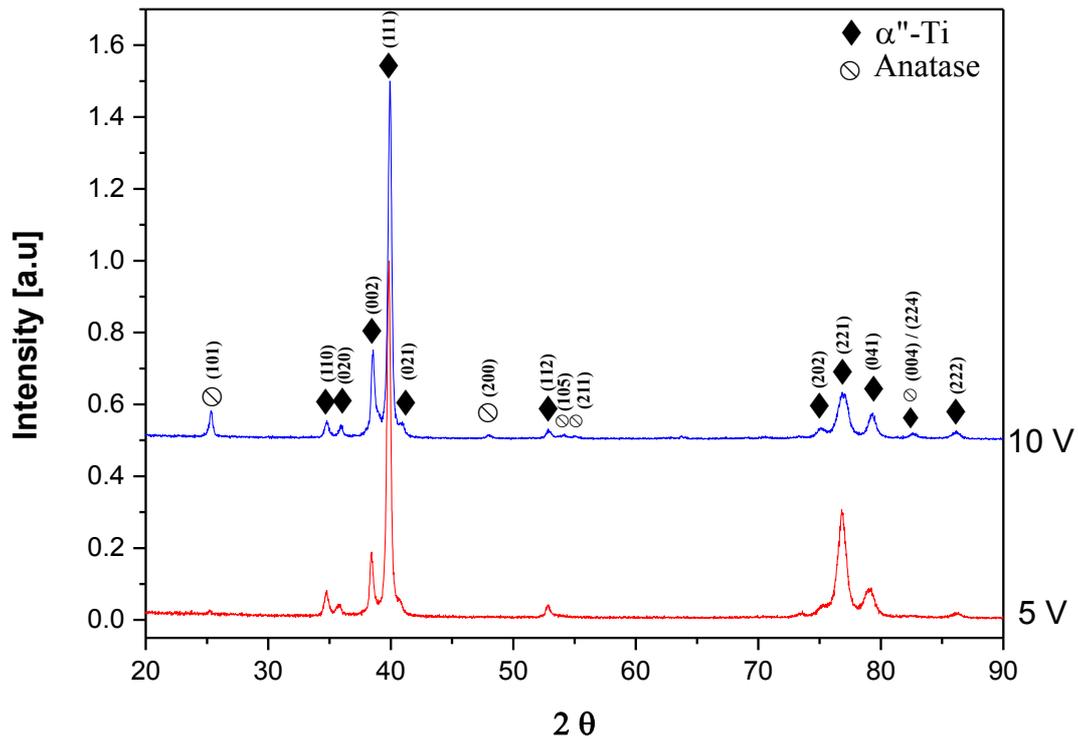


Fig. 2 - X-ray diffraction patterns for the samples anodized Ti-7.5Mo - 5V-24h-450°C and 10V-24h-450°C

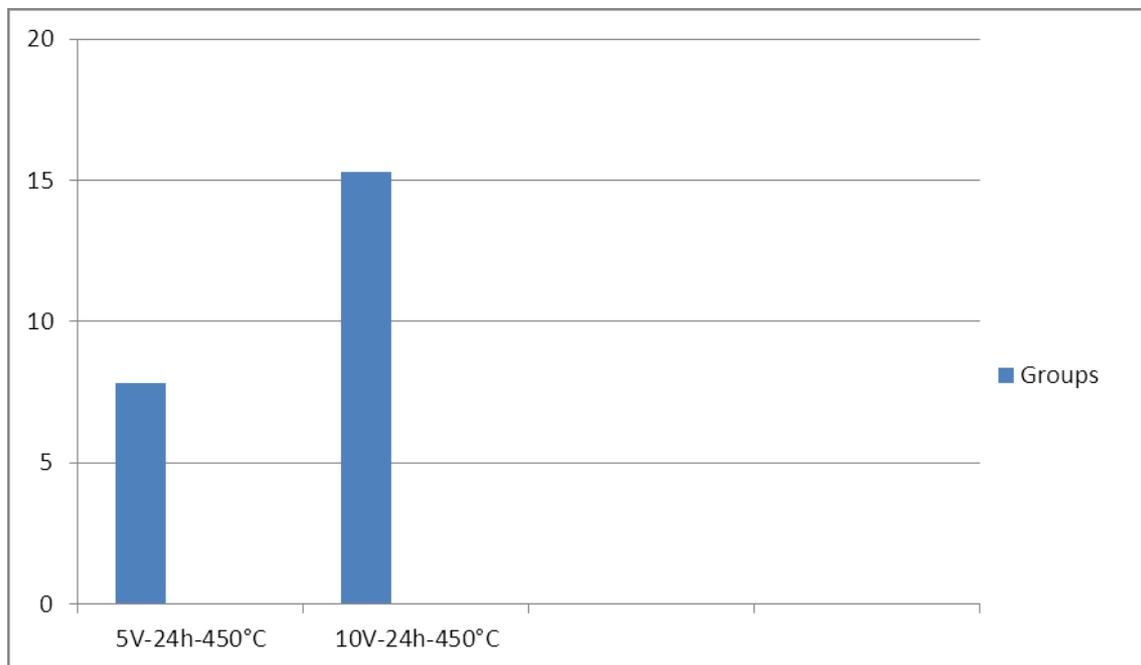


Fig. 3 - Contact angle measurements

## Conclusion

In this study, the effects of anodization potential difference applied on the formation and dimensions of the titania nanotubes were investigated. Self-organized porous nano-tubular titania was formed anodically on Ti-7.5Mo from electrolyte containing glycerol and 0.25%  $\text{NH}_4\text{F}$ .

It was found that anodization potential contributed mainly to tuning the nanotube diameter. The increasing of the anodization potential increased the nanotube diameter and the contact angle.

For samples anodized at 5 V the nanotube layer is amorphous and it is crystalline to samples anodized at 10 V. In this study, the anatase phase is present only in samples anodized at 10V, and its appearance is probably due to the greater diameter of the TiO<sub>2</sub> nanotubes formed during the application of this potential, making this to the better condition for biomedical applications. The anatase oxide phase is essential for increasing osseointegration because amorphous surfaces can be easily dissolved in body fluids.

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