

Cross-linked PEDOT: PSS as an alternative for low-cost solution-processed electronic devices

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ABSTRACT

A number of solution-processed electronic materials have been developed for application in printed devices, which are normally built by piling up several coated layers one on the top of the other. To achieve low-cost and simple to manufacture printed devices, stable and easy to process layers are mandatory. Here we introduce a mixture of poly(3,4-ethylenedioxythiophene):polystyrene sulfonate (PEDOT:PSS) with 3-glycidoxypropyl-trimethoxysilane (GPTMS) as solution-processed material that becomes insoluble in water after film processing. We performed morphological, optical and electrical characterization in the proposed composite and highlight its advantages as a stable transparent electrode in light-emitting diodes as well as in electrochromic devices.

1. Introduction

Soluble electronic materials have been profusely investigated in the past few years and are prone to become the system of choice in the new era of flexible electronics. The combination of easy solution processability with key electronic properties have made them great candidates for large-area and low-cost devices [1–5]. For instance, soluble materials have been successfully employed as active material in light-emitting devices, such as polymer light-emitting diodes (PLEDs) [6] and polymer light-emitting electrochemical cells (PLECs) [7], to mention a few. Recently, a new light-emitting composite was proposed, comprising a conducting polymer and an inorganic light-emitting powder to efficiently produce light-emitting devices, (known as LECEL – Light-Emitting Composite Electroluminescent Device) [8]. It is well-known that light-emitting devices have to be fabricated with, at least, one transparent electrode to allow light to exit the cell efficiently. The most used transparent electrode in light-emitting devices is the indium tin oxide (ITO), which is normally coated by sputtering techniques [9,10], and offers sheet resistance range of 10–100 Ω/sq and transmittance higher than 70% (at $\lambda = 550$ nm).

One of the main challenges when producing light-emitting devices is the need of stacking several layers on the top of the other. In the simplest structures, it is necessary to intercalate the active layer between a

high work function electrode and a low work function electrode to produce PLEDs [6,11]. However, in order to achieve highly-efficient emission from PLEDs, more complicated architectures are needed, which may involve the addition of an electron and/or hole transport layers between the electrodes and the active layers [12]. On the other hand, light-emitting devices like PLECs and LECELs present less restrictions regarding balanced electron/hole injection from the electrodes [7,8,13,14], dispensing the use of work-function matched electrodes and making them appropriate candidates for the use of solution-processed electrodes. Recently, for instance, PLECs using graphene as bottom electrode and poly(3,4-ethylenedioxythiophene):polystyrene sulfonate (PEDOT:PSS) as top electrode have been proposed [15]. Along the same line, Liang et. al produced PLECs using silver-nanowires/single-walled-carbon-nanotubes bilayer as bottom electrode and silver-paste as top electrode [16]. These recent discoveries pushed the development of solution processed electrodes, mainly using hot materials, such as graphene, carbon-nanotubes and metallic nanostructures as active materials [15,17–27].

Reduced graphene, carbon nanotubes and metallic nanostructures are great candidates for the fabrication of semitransparent electrodes due to its highly-conductivity and chemical stability properties. However, these materials present high optical absorption in the visible range, demanding ultra-thin films to achieve high optical

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transmittance. Usually, ultra-thin films are obtained by deposition techniques like spin-coating, which are not compatible with large scale production methodologies. An alternative is the use of solution-processed electrodes obtained via inkjet printing or spray-coating deposition, using semitransparent conducting polymer such as PEDOT:PSS [17,28]. However, being a water-soluble polymer, PEDOT:PSS based compounds are not chemically stable, especially when exposed to moisture and aqueous solution. However, on the other hand, organic/silica hybrid materials, also known as organic modified silicates (ORMOSIL), can be prepared as high-transparent colloids, which can result in a wide variety of condensate materials, yet insulating, with high chemical stability and outstanding mechanical and optical properties; not to mention its crosslinking nature [29,30].

It is well-known that water-stability and adhesion are the major problem of PEDOT:PSS film when used as conductive electrode or sensing media. Due to easy swelling, cracking-off, and dissolution of PEDOT:PSS film in water, researchers normally add cross-linkers, such as ORMOSIL, that holds the film integrity in place when interfacing PEDOT:PSS films with water solution or even when exposed to humidity. For instance, the Organic Electrochemical Transistor (OECT) and biosensing community have been using silanes for years to stabilize PEDOT:PSS film when interfacing it with biological environment [31–34].

Here we propose a new composite, which can be applied to light-emitting devices as semitransparent electrode, as well as the active layer on electrochromic devices and to produce phosphors-based electroluminescent materials. The main objective is to combine the high transparency and chemical stability of ORMOSIL materials with the enhanced conductivity of PEDOT:PSS, focusing in large scale production. We have used 3-glycidoxypropyltrimethoxysilane (GPTMS) as source of the ORMOSIL material and studied the optoelectronic and morphological properties of composites with different weight ratios between PEDOT:PSS and GPTMS. We observe different regimes of conductivity with varying GPTMS/PEDOT:PSS ratio, and provide guidelines on how conductivities and film transparency can be selected by varying the GPTMS/PEDOT:PSS ratio in the mixture as well as the composite thicknesses.

2. Material and methods

The highly conductive PEDOT:PSS ink was purchased from Clevios™ under the name of PH-1000. Films conductivity of up to 1000 Scm^{-1} were produced after addition of 5%-wt of ethylene glycol to the solution [35]. The GPTMS used in this work was provided by Sigma-Aldrich and used as is, whereas the Mn-doped zinc silicate was purchased from Fluka. The molecular structures of both materials are shown in Fig. 1. PEDOT:PSS/GPTMS solutions with PEDOT:PSS concentrations varying from 10%-wt up to 100%-wt (pure PEDOT:PSS) were prepared and stirred for 24 h at room temperature. The solutions were used to obtain films by spray coating, using a manual aerograph (Western, model AER-4), onto clean glasses substrates on top of a hot plate at 40°C (303 K). A mechanical shadow mask forming films with 9.0 mm width was used to pattern the PEDOT:PSS film. Subsequently, the films were baked at 373 K for 15 min to eliminate solvent (water) traces. Top Au electrodes were deposited in high vacuum ($< 10^{-6}$ mbar) by thermal evaporation. Electroluminescent devices were obtained with the active layer, comprising GPTMS/PEDOT:PSS/ $\text{Zn}_2\text{SiO}_4\text{:Mn}$ (0.045/0.005/0.95%-wt.), deposited by drop-casting the precursor solution onto either ITO-coated glass substrates ($50 \Omega/\text{sq}$ sheet resistance) or glass substrates coated by a GPTMS/PEDOT:PSS (50/50%-wt) layer. After spreading the active layer solution onto the substrates, the devices were left to rest inside a furnace for 24 h, at room temperature, to allow the silicate to sedimentate. Subsequently, the temperature was increased to 80°C (353 K) and left for 24 h for solvent evaporation. All films were obtained without pinholes and with good edge definition, as presented in Fig. S1 to S4 in the supplementary information (SI). The morphological

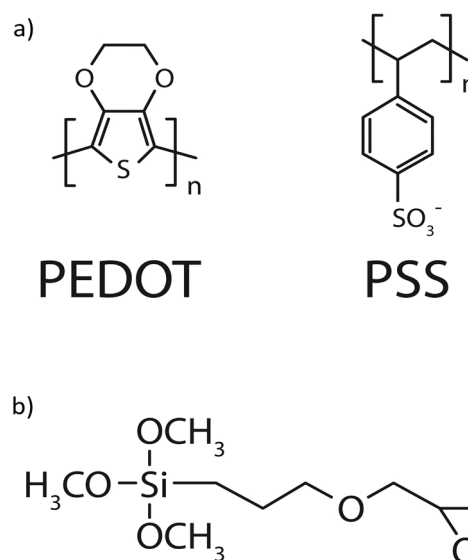


Fig. 1. Chemical structure of a) poly(3,4-ethylenedioxythiophene):polystyrene sulfonate (PEDOT:PSS) and b) 3-glycidoxypropyltrimethoxysilane (GPTMS).

characterization of the obtained films by optical microscopy, profilometry and atomic force microscopy are also presented in the SI.

3. Results and discussion

Fig. 2(a) and (b) present the dependence of conductivity of GPTMS/PEDOT:PSS films on PEDOT:PSS content in the mixture (monolog and linear scales, respectively). Two regimes of conductivity can be observed in the curve, with a clear transition at 40%-wt of PEDOT:PSS. Below 40%-wt, we observe an exponential dependence of film conductivity with PEDOT:PSS content, whereas above 40%-wt the conductivity scales linearly with the PEDOT:PSS amount. This is due to the formation of percolation paths of conductive PEDOT:PSS aggregates throughout the GPTMS matrix. For low PEDOT:PSS concentrations, the conductive aggregates do not overlap efficiently and the overall conductivity of the film is limited by hopping of mobile charges between spatially separated aggregate regions, which reduces the overall film conductivity. An evidence of the disconnected PEDOT:PSS-rich aggregates was observed in a sample containing 15%-wt of PEDOT:PSS using the near-field FTIR spectroscopy microscopy, presented in Fig. S5 from supplementary information. For electrode applications, it is mandatory to be on the higher concentration regime, where percolation paths are established and conductivities are higher. However, for several other applications, such as phosphors-based electroluminescent materials, a lower overall conductivity might be needed, which can be achieved by selecting a lower PEDOT:PSS concentration in the mixture as will be discussed in the next sections.

Fig. 2(c) shows the dependence of the optical absorption coefficient of GPTMS/PEDOT:PSS films on PEDOT:PSS concentration. The shape of the absorption curve does not vary considerably with the addition of GPTMS in the composite, due to the high transparency of GPTMS in the visible spectrum. Given that the PEDOT:PSS is the only optically active material in the composite (within the analyzed energy range), the major observed change is a linear increase on the absorption intensity with increase of PEDOT:PSS content. This is better observed in Fig. 2(d), where the absorption coefficient is plotted for three different wavelengths: 599, 555 and 446 nm (in the blue, green and red ranges of the visible spectrum). Fig. 2 as a whole forms the basis to build a figure of merit that relates the sheet resistance, light transmittance and cost of production of the conductive composite here presented. The results are summarized in Fig. 3(a). A similar curve was already published by Carr et. al. and further details on its construction can be found in reference

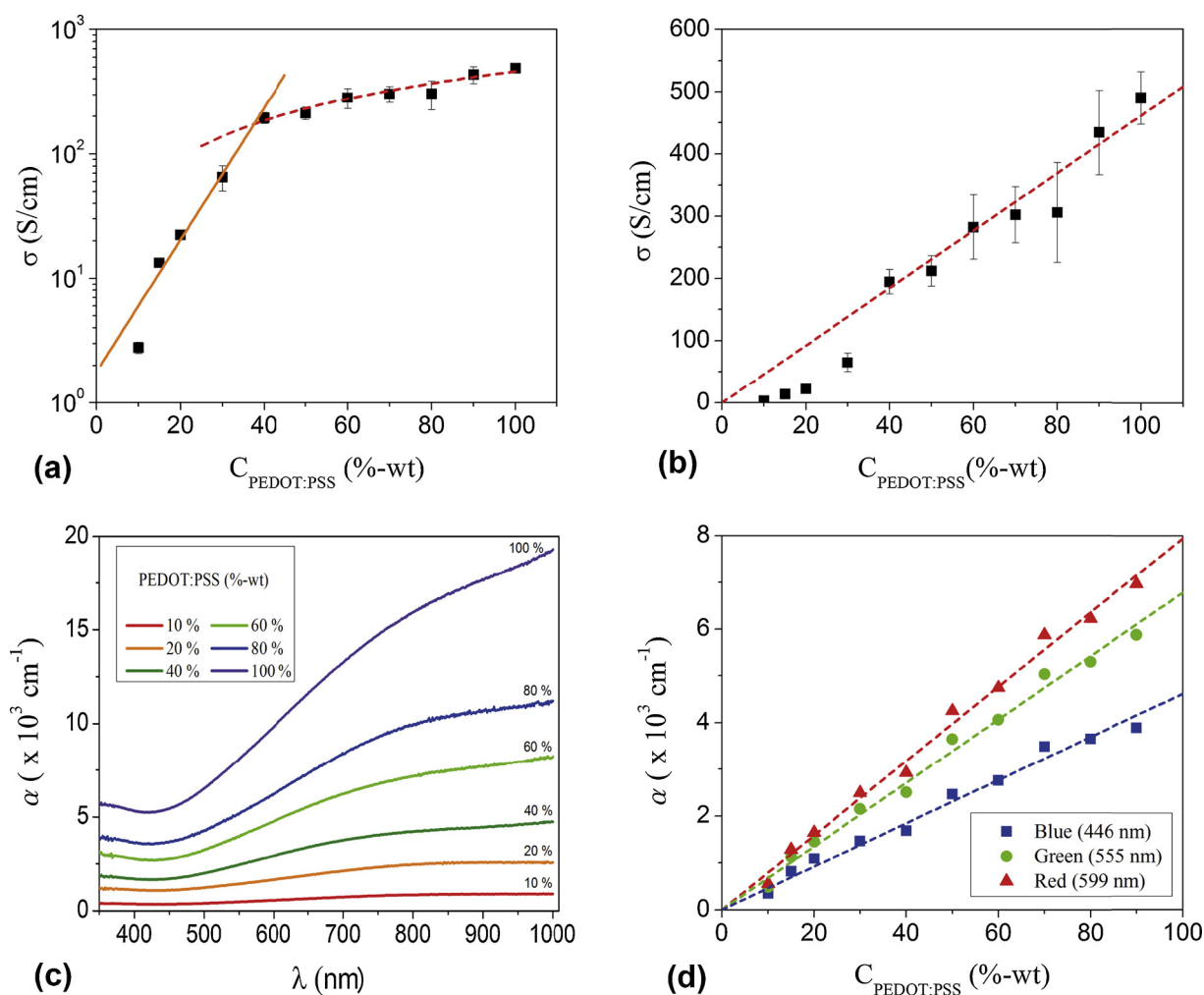


Fig. 2. Electrical conductivity and optical absorption coefficient as function of the PEDOT:PSS weight concentration in GPTMS/PEDOT:PSS mixtures. (a) Electrical conductivity in monolog scale; (b) Electrical conductivity in linear scale; (c) Absorption coefficient as function of the incident light wavelength; (d) Absorption coefficient as function of the PEDOT:PSS weight ratio at specific wavelengths.

[28]. The diagram in Fig. 3(a) evidences the limits in optical transmittance and sheet resistance, with dashed lines representing the film characteristics at a specific film thicknesses, and solid lines representing the behavior at constant PEDOT:PSS concentrations. This figure of merit is suitable to dictate guidelines to produce films of GPTMS/PEDOT:PSS for application as semitransparent electrodes. For reference, we also added on the chart the mean position of commercial ITO (Sigma-Aldrich Indium tin oxide, reference number 639281 and 576352). Here we are not considering the production cost of an ITO film, as it would be way far out in the current chart. The commercial ITO sits almost on the top of 0.5 μm thickness and 50%-wt PEDOT:PSS reference curves. Therefore, similar ITO electrodes performance can be achieved using the strategy reported here at a lower production cost. Additionally, in the supplementary information (Fig. S6) we present figures of merit obtained for samples produced with a mixture of PEDOT:PSS and GPTMS:tetraethyl orthosilicate (TEOS) as the transparent phase and cross-linker agent.

It is worthwhile mentioning that the addition of GPTMS in PEDOT:PSS solution also enhanced the film stability in aqueous environment. Fig. S7 presents the changing of relative absorbance over time of water-soaked pure PEDOT:PSS and crosslinked GPTMS/PEDOT:PSS films. Normally, a decrease in the relative absorption is associated with losses in the film mass. It can be seen from Fig. S7 that pure PEDOT:PSS film dissolves in a matter of minutes when immersed in water. However, for crosslinked GPTMS/PEDOT:PSS the relative

absorption remains almost unvaried, indicating high stability of the films in aqueous environment. The most accepted cross-linking reaction between PEDOT:PSS and GPTMS occurs between the sulfonate groups ($-\text{SO}_3^-$) present in the excess of PSS and the epoxy ring of GPTMS. As for the methoxysilane group found also in the GPTMS, they are more likely to bond either to the methoxysilane group of another GPTMS molecule or to the glass substrate [36].

As a proof of concept, we connected two 3 V batteries in series with a LED and a film 10 x 10 cm GPTMS/PEDOT:PSS (50/50%-wt) onto a glass substrate. The picture in Fig. 3(b) illustrates the conductivity and transparency that can be achieved with the current strategy when producing GPTMS/PEDOT:PSS electrodes. To show the flexibility of the films, Fig. 3(c) displays a picture of a film coated onto polyethylene terephthalate (PET) substrate in front of a small light bulb.

In order to test such material in practical applications, we experimented them as component in a phosphors-based electroluminescent material, as shown in Fig. 4. Devices were produced by mixing the GPTMS/PEDOT:PSS (90/10%-wt) composite as conductive matrix with $\text{Zn}_2\text{SiO}_4\text{:Mn}$ microparticles as light-emitting phase. Even though the mechanisms to convert the electric energy to light in this class of materials are not completely understood, functional devices were obtained using non-percolated conductive matrix [8]. Due to the requirements for producing such EL device we used a GPTMS/PEDOT:PSS (90/10%-wt) of GPTMS/PEDOT:PSS material, which is below the percolation point. The current vs. voltage and luminance vs. voltage curves, in

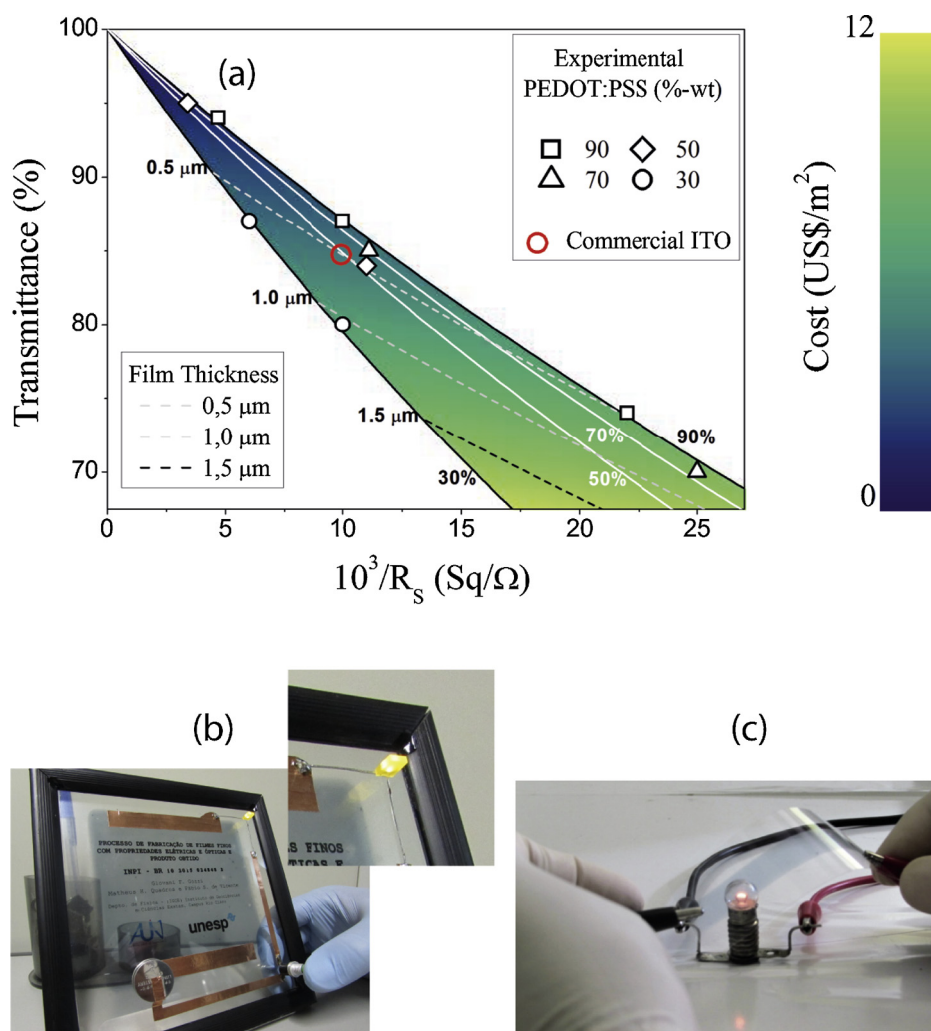


Fig. 3. (a) Figure of merit for transparent and conductive films produced with GPTMS/PEDOT:PSS mixtures; (b) picture showing the proof of concept using a series circuit with a LED, two batteries (3 V each) and the transparent film as resistor; (c) picture showing the flexibility of a film coated over PET substrate.

Fig. 4(a), show that the device produced using ITO and gold as electrodes, with a 140 μm thick active layer and active area of 0.85 cm², operates on both polarities, with turn-on voltages in the 30–36 V, which is significantly low, considering the film thickness (electric field of 2.6×10^5 V/m). These characteristics are quite similar to that published on a similar light-emitting device [8,37–43]. The device luminance reached 2500 cd/m² and a luminous efficacy of 26 cd/A, as can be seen in Fig. 4(a) and (b). As illustrated by Fig. 4(c) and 4(d), the device emitted green light ($\lambda = 535$ nm), producing a CIE 1931 Illuminant E coordinates of $x = 0.222$ and $y = 0.692$ and a color purity of 85%. Usually devices produced with light-emitting composites comprising inorganic microparticles exhibit maximum luminance of 50 cd/m² [37,38] and 100 cd/m² [8], and it is necessary a complex device structure to achieve numbers close to 10^3 cd/m² [39]. Devices with maximum luminance in the range 10^3 – 10^4 cd/m² were already produced using ultrathin films of composites comprising nanoparticles [40–43]. Although the light intensity is fairly high on these devices, their luminous efficacy are very low, normally not higher than 10 cd/A.

We have also explored the GPTMS/PEDOT:PSS films as active material in an electrochromic device. PEDOT:PSS is known as a mixed conductor, being able to support both electronic and ionic conduction. A major application of this characteristic is the use of PEDOT:PSS as the active layer of organic electrochemical transistor (OECTs) as well as an electrochromic device [44,45]. When ions enter the materials, polaronic states are formed within the material's bandgap changing its

optical properties, causing a change on the film coloring. The advantage of PEDOT:PSS over other electrochromic materials is due to its good response time, great color contrast and, more importantly, its ability to be processed from a water emulsion. The electrochromic device was produced by coating a layer of GPTMS/PEDOT:PSS (10/90%-wt) on top of an ITO electrode. The film was immersed in an electrolyte solution (100 mM NaCl). An Ag/AgCl reference electrode was placed in the solution and a potential was applied between the Ag/AgCl and the ITO electrode. Fig. 5(a) shows the absorption spectra for two voltages. With a null voltage between the ITO and the reference electrode, the GPTMS/PEDOT:PSS (10/90%-wt) is barely transparent. However, when a positive voltage is applied to the Ag/AgCl, Na⁺ ions penetrates the composite, changing its absorption properties, as can be seen by the absorption peak for + 1 V (solid line). Fig. 5(b) and (c) are real pictures of the experimental setup when polarizing the film with 0 V and + 1 V respectively: the latter shows a dark-blue color, consistently with the absorption spectrum.

4. Conclusion

In summary, we demonstrated a new strategy to produce stable PEDOT:PSS films to be used as semitransparent electrodes, active matrix in light-emitting device and electrochromic devices. The film conductivity measurements show a clear indicative of a phase separation of the GPTMS and PEDOT:PSS regions occurring in the nanoscale regime,

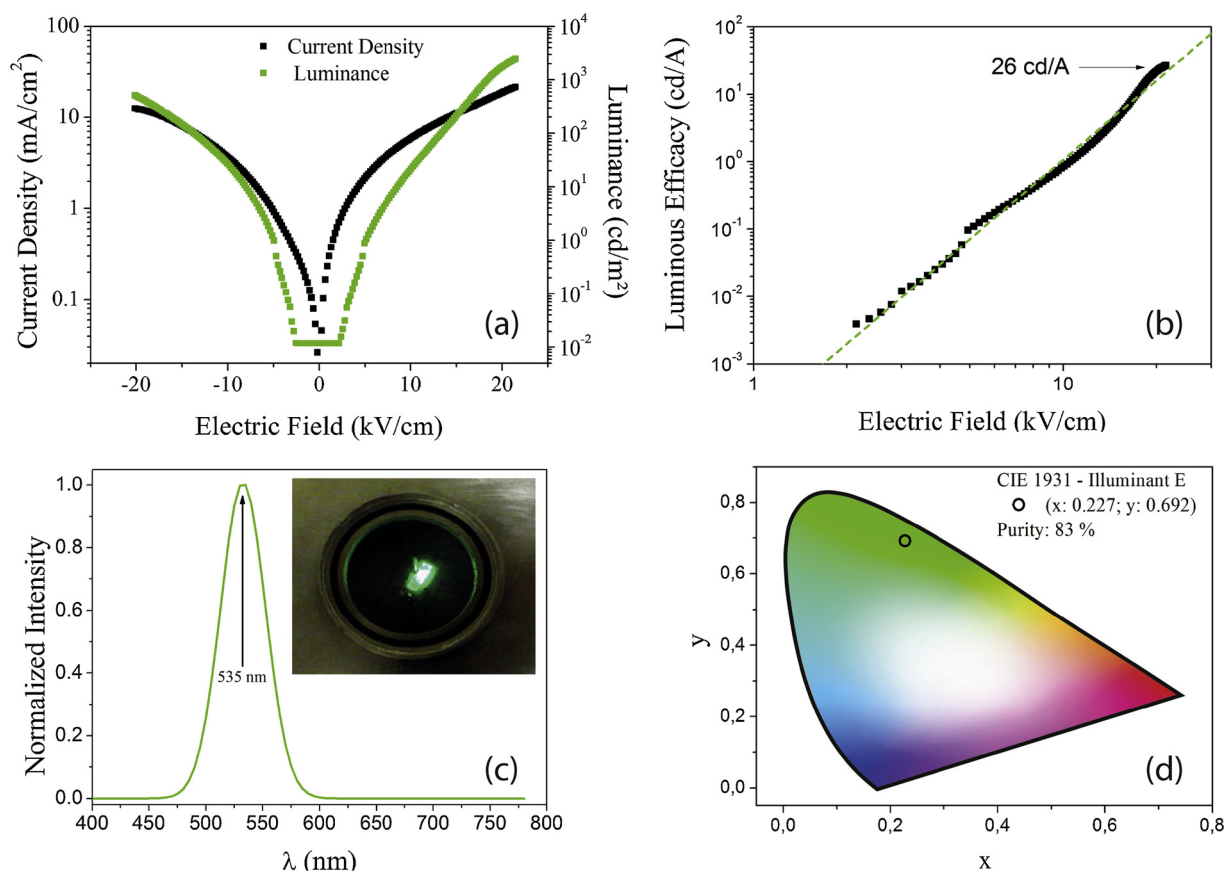


Fig. 4. Results obtained of light-emitting devices produced with the composite GPTMS/PEDOT:PSS/Zn₂SiO₄:Mn as active material: (a) Current density vs. electric field and luminance vs. electric field characteristics curves, (b) luminous efficacy vs. electric field, (c) electroluminescence spectra of a device produced with ITO and Gold electrodes; (d) CIE color diagram.

i.e., the films are homogeneous in macroscopic and microscopic scales. As consequence of the nanoscale phase separation, films produced with less than 40%-wt of PEDOT:PSS exhibits an exponential dependence on the PEDOT:PSS concentration, which is characteristic of systems below the percolation limit of the conductive phase. On the other hand, films produced with concentrations above 40%-wt of PEDOT:PSS, presented a linear dependence of the conductivity on the conductive phase concentration, evidencing the achievement of the percolation limit. The

electrical conductivity of the mixture varied between 2 S/cm (10%-wt of PEDOT:PSS) and 600 S/cm (90%-wt of PEDOT:PSS). As proof of concept, we applied those composites as conductive matrix to a phosphors-based electroluminescent device as well as to electrochromic systems.

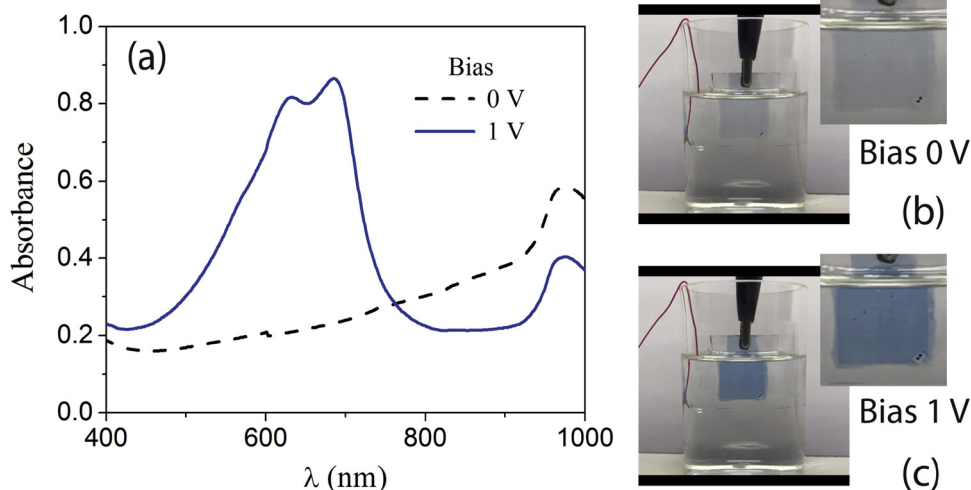


Fig. 5. Results obtained from an electrochromic device produced with GPTMS/PEDOT:PSS as active material and water solution of NaCl (100 mM) as electrolyte: (a) device absorbance spectra for 0 V and 1 V bias; (b) picture of the device biased at 0 V; (c) picture of the device biased at 1 V.

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Appendix A. Supplementary data

Supplementary material related to this article can be found, in the online version, at doi:<https://doi.org/10.1016/j.synthmet.2018.04.002>.

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