



UNIVERSIDADE ESTADUAL PAULISTA
"JÚLIO DE MESQUITA FILHO"
Campus de São José dos Campos
Instituto de Ciência e Tecnologia

TABATA DO PRADO SATO

**DESENVOLVIMENTO DE BIOMATERIAIS À BASE DE
QUITOSANA: matriz de fibras eletrofiadas para regeneração
tecidual e de hidrogel coacervado para entrega controlada de
fármaco**

2019

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Tese apresentada ao Instituto de Ciência e Tecnologia, Universidade Estadual Paulista (Unesp), Campus de São José dos Campos, como parte dos requisitos para obtenção do título de DOUTOR, pelo Programa de Pós-Graduação em ODONTOLOGIA RESTAURADORA.

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Orientador: Prof. Ass. Alexandre Luiz Souto borges

Coorientador: Prof. Dr. Artur José Monteiro Valente

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À ciência brasileira.

À educação brasileira.

Ao povo brasileiro, que financiou esta pesquisa.

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*" Eu sou o senhor de meu destino; Eu sou o capitão de minha alma. ". William
Ernest Henley*

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RESUMO

Os atuais avanços no desenvolvimento de biomateriais caminham para 2 áreas promissoras: a de regeneração tecidual e a de entrega controlada de fármacos. Assim, o presente estudo objetivou a síntese e caracterização de diferentes matrizes (fibras e hidrogel) à base de quitosana, a fim de se obter materiais biomiméticos para atuação em ambas áreas. Para regeneração, delineou-se a síntese de um arcabouço de fibras de quitosana com e sem cristais de nanohidroxiapatita onde, para isso, foram eletrofiadas soluções de quitosana (Ch) e de quitosana com nanohidroxiapatita (ChHa). Os espécimes de Ch apresentaram maior homogeneidade e maior diâmetro médio de fibras (690 ± 102 nm) que ChHa (358 ± 49 nm). No teste de viabilidade celular e na atividade da fosfatase alcalina não houve diferença estatística entre os grupos experimentais (Ch e ChHa), porém ambos diferiram do grupo controle ($p < 0,001$). Para o âmbito de liberação de fármacos, sintetizou-se, pela técnica de emulsão, dois tipos de hidrogéis: o primeiro, uma mistura da fase aquosa da solução de Ch (1 mL) e da solução de DNA (1 mL) (1:1) e o segundo, mistura da fase aquosa da solução de Ch (1 mL) e solução de Pectina (1 mL) (1:1). Ambas misturas foram realizadas em álcool benzílico (5 mL) com instrumento de dispersão de alto desempenho ($31-34000$ rpm min^{-1} por 5 min). Após a obtenção dos géis, 20mg de cada grupo foram imersos em uma solução aquosa de Própolis Verde (PV), na concentração de $70 \mu\text{g/mL}$ por 2 h e a cinética de liberação do PV foi analisada a 25 e 37°C em água e saliva artificial. Os espécimes obtidos foram liofilizados e depois caracterizados físico-quimicamente. A presença de pectina e de DNA foi comprovada por FTIR. A sorção de PV induziu uma modificação significativa da superfície do gel. Constatou-se uma separação de fases entre a quitosana e o DNA. A eficiência do encapsulamento não mudou significativamente entre 25 e 37°C . A cinética de liberação na água ou na saliva apresentou um mecanismo de duas etapas. E os resultados biológicos exibiram que esses materiais são aceitáveis no ambiente biológico. Assim, conclui-se que a matriz de fibras de quitosana com nHAp é capaz de promover diferenciação celular e a matriz de hidrogel de quitosana com Pectina ou DNA possui potencial para a liberação controlada de fármacos.

Palavras-chave: Quitosana. Nanofibras. Hidrogel. DNA. Sistemas de Liberação de Medicamentos

Sato TP. Development of chitosan-based biomaterials: electrospun fiber matrix for tissue regeneration and coacervate hydrogel for controlled drug delivery [doctorate thesis]. São José dos Campos (SP): São Paulo State University (Unesp), Institute of Science and Technology; 2019.

ABSTRACT

Current advances in biomaterial development are moving to 2 promising areas: tissue regeneration and controlled drug delivery. Thus, the present study aimed the synthesis and characterization of different matrices (fibers and hydrogel) based on chitosan, in order to obtain biomimetic materials for performance in both areas. For regeneration, the synthesis of a scaffold of chitosan fibers with and without nanohydroxyapatite crystals was delineated, where chitosan (Ch) and chitosan with hydroxyapatite (ChHa) solutions were electrospun. Ch specimens presented higher homogeneity and larger mean fiber diameter ($690 \pm 102 \text{ nm}$) than ChHa ($358 \pm 49 \text{ nm}$). In the cell viability test and alkaline phosphatase activity there was no statistical difference between the experimental groups. (Ch and ChHa), but both differed from the control group ($p < 0,001$). For the drug release scope, two types of hydrogels were synthesized by the emulsion technique: the first, a mixture of the aqueous phase of Ch solution (1 mL) and DNA solution (1 mL) (1:1) and the second, mixture of the aqueous phase of the Ch solution (1 mL) and Pectin solution (1 mL) (1:1). Both mixtures were performed in benzyl alcohol (5 mL) with high performance dispersion instrument ($31\text{-}34000 \text{ rpm min}^{-1}$ for 5 min). After obtaining the gels, 20mg from each group were immersed in an aqueous solution of Propolis Green (PV), at a concentration of $70 \mu\text{g/mL}$ for 2 h and the release kinetics of PV were analyzed at 25 and 37°C in water and artificial saliva. The obtained specimens were lyophilized and then physically-chemically characterized. The presence of pectin and DNA was confirmed by FTIR. PV sorption induced a significant modification of the gel surface. A phase separation was found between chitosan and DNA. Encapsulation efficiency did not change significantly between 25 and 37°C . The release kinetics in water or saliva presented a two-step mechanism. And the biological results showed that these materials are acceptable in the biological environment. Thus, it is concluded that the nHAp chitosan fiber matrix is capable of promoting cell differentiation, whereas the chitosan hydrogel matrix with Pectin or DNA are potential biomaterials for controlled drug release.

Keywords: Chitosan. Nanofibers. Hydrogel. DNA. Drug Delivery Systems.

1 INTRODUÇÃO

A Engenharia de Tecidos (ET) oferece perspectivas promissoras para as áreas de medicina e odontologia atuais, pois visa regenerar tecidos danificados, com materiais biomiméticos que restauram ou conservam a função do tecido vivo (Atala, 2004; Bonassar, Vacanti, 1998).

Através das técnicas de nanofabricação, a ET desenvolve ambientes favoráveis para o desenvolvimento celular, arcabouços com propícias características de adesão de proliferação (Kingsley et al., 2013; Zhang et al., 2018). Já a liberação controlada de fármacos surge no contexto de interações de tecidos com fluidos extracelulares e células especializadas (Chung et al., 2007).

Nanoestruturas poliméricas apresentam características interessantes que suportam uma ampla gama de novas aplicações biotecnológicas (Sandreschi et al., 2016). Neste contexto, uma atenção crescente tem sido dada aos polímeros naturais (Ngwuluka et al., 2016; Titorencu et al., 2017), como os polissacarídeos, devido à sua abundância na natureza e a um apelo global por processos sustentáveis (Honarkar, Barikani, 2009).

Na odontologia, a utilização de nanomateriais já é uma realidade do cenário clínico, como ocorre com dispositivos para diagnóstico de câncer bucal e sistemas de administração de fármacos para interromper a formação de biofilme (AlKahtani, 2018).

Vários são os métodos para fabricação desses materiais, como o *drawing*, uma extrusão por meio de uma secagem giratória (Ondarçuhu, Joachim, 1998; Xing et al., 2008), a síntese *template*, que utiliza uma membrana nanoporosa como um molde de fibras (Feng et al., 2002; Martin, 1996) e a automontagem (*self-assembly*), uma construção molécula por molécula de estruturas bem definidas (Hiraoka, 2015; Sun et al., 2015). Entretanto, a técnica de eletrofiação

parece ser uma técnica relativamente mais simples e efetiva para a produção de fibras com uma variedade de polímeros de morfologia controlável (Frenot, Chronakis, 2003; Reneker, Chun, 1996; Shanmuga Sundar, Sangeetha, 2012).

Nesse contexto, fibras poliméricas eletrofiadas apresentam propriedades interessantes para muitas aplicações biomédicas, tais como elevada área superficial de contato quando comparadas com outros materiais utilizados nesta área (Huang et al., 2003; Kumar et al., 2016; Mehrasa et al., 2015; Wu et al., 2015).

O processo de eletrofiação é composto por um campo elétrico criado por uma alta tensão aplicada a uma seringa com solução polimérica ou polímero fundido, um capilar metálico e um coletor. Quando a solução é bombeada através de uma bomba de infusão, ou por força gravitacional, atravessa a seringa acoplada ao capilar (eletrodo positivo) e quando o campo elétrico excede a tensão superficial da solução, jatos poliméricos são ejetados em direção ao coletor (eletrodo negativo), posteriormente gerando fibras sólidas devido à evaporação do solvente (Ahmed et al., 2015; Lyons et al., 2004).

Atualmente, biopolímeros naturais são amplamente utilizados em diversas áreas e com várias aplicações alavancadas por métodos como o da eletrofiação (Sell et al., 2010; Soares et al., 2018). Assim, a quitosana, um biopolímero obtido pela desacetilação da quitina, que está presente nos exoesqueletos de crustáceos, insetos, moluscos e parede celular de fungos (Gopalan Nair, Dufresne, 2003; Ifuku, 2014) apresenta, nos últimos tempos, fibras efetivamente eletrofiadas a partir de soluções em ácido acético (Geng et al., 2005; Homayoni et al., 2009) ou usando ácido trifluoroacético (TFA), que pode ser associado com diclorometano (DCM) (Ohkawa et al., 2004).

Estudos recentes sobre a caracterização estrutural das fibras de quitosana (Karpova et al., 2016) apontaram para uma ampla gama de aplicações, desde estabilidade de insulina oral (Chen et al., 2015), regeneração óssea guiada

(Foster et al., 2015; Ma et al., 2014; Qasim et al., 2017), defeitos ósseos alveolares até mesmo nanocomplexos microbicidas para proteção contra infecções por HIV (Boyapalle et al., 2015).

Além disso, muitos estudos associaram a quitosana a outros componentes, como nanopartículas de prata (Hadipour-Goudarzi et al., 2014; Liu et al., 2015), colágeno (Chen et al., 2007; Chen et al., 2010) e extratos naturais (Charernsriwilaiwat et al., 2013) para eletrofiação fibras com diferentes características, tais como a atividade antibacteriana (Hadipour-Goudarzi et al., 2014; Liu et al., 2015), cicatrização alternativa de feridas (Charernsriwilaiwat et al., 2013) e para ambientes adequados à viabilidade celular (Chen et al., 2007; Chen et al., 2010). A hidroxiapatita, importante componente da fase mineral óssea, é um nanomaterial que pode ser incorporado em fibras de quitosana para atividade de regeneração óssea (Kong et al., 2006). Além disso, sob nanoescala, a nanohidroxiapatita (nHAp) possui alta aplicabilidade desde dentifrícios anti-erosão (Aykut-Yetkiner et al., 2014) até engenharia tecidual (Nga et al., 2014; Zhou, Lee, 2011).

Neste estudo, sob essa escala nanométrica, a nHAp foi obtida (Barbosa et al., 2013) para ser incorporada nas fibras de quitosana devido à sua semelhança com a hidroxiapatita do tecido ósseo, reduzindo as limitações biomiméticas.

Além das estruturas fibrilares poliméricas, o presente estudo contemplou a síntese e caracterização de um novo composto de sob uma matriz de hidrogel. Os hidrogéis são conhecidos como redes 3D poliméricas reticuladas, capazes de absorver e reter grandes quantidades de água ou fluidos biológicos (Ma et al., 2016; Peppas, 1986), possuindo uma microestrutura semelhante à matriz extracelular (MEC) (Ma et al., 2016) e significativa biocompatibilidade (Gong, 2006).

Sendo assim, aproveitando a carga positiva do polímero central estudado, a quitosana, decidiu-se investigar a formação de hidrogéis de quitosana-DNA e quitosana-pectina coacervados.

O ácido desoxirribonucleico (DNA) é um biopolímero composto de açúcar desoxirribose, capaz de transportar informações genéticas (Uzumcu et al., 2016). Os hidrogéis de DNA têm propriedades únicas, como biocompatibilidade, ligação seletiva e reconhecimento molecular (Ishizuka et al., 2006; Murakami, Maeda, 2005), criando uma gama de aplicações na liberação controlada de fármacos (Costa et al., 2019; Costa et al., 2011) e engenharia de tecidos (Costa et al., 2014; Li et al., 2013; Um et al., 2006). De fato, entre os polieletrólitos biológicos, o DNA sempre atraiu interesse particular, e existem numerosos estudos sobre as interações entre o DNA e os polímeros (Costa et al., 2015; Costa et al., 2018; Jorge et al., 2010; Jorge et al., 2013). A formação de hidrogéis coacervados é alcançável quando o DNA é completamente ionizado em pH neutro (Bravo-Anaya et al., 2016a) permitindo a interação com moléculas carregadas positivamente, como a quitosana (Bravo-Anaya et al., 2016b).

Por outro lado, a pectina (P) um polissacarídeo, originário da parede celular de plantas, é usado como polímero gelificante e estabilizante em alimentos e produtos biomédicos devido à influência na saúde humana, como níveis de colesterol e sangramento local (Mohnen, 2008; Thakur et al., 1997), além disso, também é capaz de formar hidrogéis coacervados com quitosana (Cesar Filho et al., 2018; Neufeld, Bianco-Peled, 2017).

Essas duas matrizes de hidrogel (CS-DNA e CS-Pec) foram analisadas no presente estudo na liberação controlada de Própolis Verde (GP), um fitoterápico produzido pelas abelhas *Apis mellifera*, de *Baccharis dracunculifolia* DC (Asteraceae), usado como medicamento anti-inflamatório (de Barros et al., 2007), antifúngico (Santos et al., 2005) e antioxidante (Guimarães et al., 2012;

Nakajima et al., 2007) devido aos seus altos níveis de ácidos fenólicos (Bankova et al., 2000).

Sendo assim, propôs-se neste estudo, a síntese e caracterização de diferentes matrizes (fibras eletrofiadas e hidrogel) à base de quitosana, a fim de se obter biomateriais clinicamente complementares na regeneração tecidual e liberação controlada de fármacos.

2 ARTIGOS

2.1 Artigo – Sato TP, Rodrigues BVM, Mello DCR, Machado JPB, Bottino MC, Vasconcellos LMR, Lobo AO, Borges ALS. Potencial de carregamento de nanohidroxiapatita em fibras de quitosana eletrofiadas: estudos biológicos e morfológicos / *Potential of nanohydroxyapatite loading on electrospun chitosan fibers: biological and morphological studies**

RESUMO

A eletrofição é uma técnica para produzir fibras poliméricas ultrafinas. A quitosana é um biopolímero usado para esse fim, que origina um biomaterial com atividade antimicrobiana. A hidroxiapatita pode ser associada a este material e, assim, foram produzidas soluções de quitosana a 7 % em peso (Ch) e quitosana / nanohidroxiapatita a 0,5 % em peso (nHAp), ambas usando ácido trifluoroacético e diclorometano como solventes. Todos os materiais eletrofiados foram caracterizados por Microscopia Eletrônica de Varredura, Perfilometria e Microscopia de Força Atômica. As propriedades físico-químicas foram avaliadas por espectroscopia no infravermelho por transformada de Fourier e análises termogravimétricas. Esses materiais também foram investigados usando experimentos celulares *in vitro*. As amostras de Ch apresentaram maior homogeneidade e maior diâmetro médio de fibras ($690 \pm 102\text{nm}$) que ChHa ($358 \pm 49\text{nm}$). No teste de viabilidade celular e na atividade da fosfatase alcalina não houve diferença estatística entre Ch e ChHa ($p > 0,001$), apenas encontrada entre grupos experimentais e grupos de controle celular ($p < 0,001$), o que confirma que os arcabouços não apresentaram efeitos citotóxicos, mas provam diferenciação celular. Assim, pode-se concluir que o nHAp influenciou nas propriedades topográficas e biológicas, mas não alterou significativamente as propriedades físico-químicas das fibras de quitosana. Curiosamente, todos os arcabouços eletrofiados indicam um bom ambiente para proliferação e diferenciação celular, o que pode abrir uma ampla gama de aplicações desses materiais, como na engenharia de tecidos.

Palavras-chave: Quitosana. Hidroxiapatita. Viabilidade celular.

ABSTRACT

Electrospinning is a technique to produce ultrathin polymeric fibers. Chitosan is a biopolymer used for this purpose, which originates a biomaterial with antimicrobial activity. Hydroxyapatite can be associate to this material and thus, chitosan 7 wt% (Ch) and chitosan/nanohydroxyapatite 0.5 wt% (nHAp) solutions were produced, both using trifluoroacetic acid and dichloromethane as solvents. All electrospun materials were characterized by Scanning Electronic Microscopy, Profilometry and Atomic Force Microscopy. The physico-chemical properties were assessed by Fourier Transform Infrared Spectroscopy and Thermogravimetry analyses. These materials were also investigated using in vitro cell experiments. The Ch samples showed greater homogeneity and a higher mean diameter ($690 \pm 102\text{nm}$) than ChHa ($358 \pm 49\text{nm}$). In the cell viability test and alkaline phosphatase activity, there was no statistical difference between Ch and ChHa ($p > 0.001$), only founded between experimental groups and cell control groups ($p < 0.001$) which confirms that scaffolds did not present cytotoxicity effects but prove cellular differentiation. Thus, it can be concluded that the nHAp influenced on the topographical and biological properties but did not changed significantly the physicochemical chitosan fibers properties. Interestingly, all electrospun scaffolds indicate a good environment for cell proliferation and differentiation, which can open a wide range of these materials applications as for tissue engineering.

Keywords: Chitosan. Hydroxyapatite. Cell Survival

1. Introduction

Polymeric nanostructures present interesting features that support a wide range of new biotechnological applications [1]. In this context, growing attention has been given to natural polymers [2, 3], such as polysaccharides, due to their abundance in nature and to a global appeal for sustainable processes [4].

There are several different methods for preparing polymeric fibers, known as drawing [5, 6], template synthesis [7, 8], phase separation [9] and self-assembly [10, 11]. However, electrospinning technique appears to be cost-

effective, reliable and relatively simple technique, which is useful for producing fibers with a polymers variety of controllable morphology [12-14].

Several polymer investigations have been dealing with nano and ultrathin electrospun fibers. Their modifications have led to enforcements expansion [15]. The electrospun fibers present interesting properties for many biomedical applications, such as a high specific surface area and flexibility compared to other materials used in this area [16-19].

Electrospinning consists basically in a high voltage applied into a system constituted by a syringe with polymer solution or melt solution, a metal capillary and a collector. The solution is pumped through an infusion pump, or by gravitational force, through the syringe coupled to capillary (positive electrode). When the electric field exceeds the surface tension, the polymer jets ejected towards the metallic collector (negative electrode), which generates fibers due to the solvent evaporation [20, 21].

Therefore, chitosan, a biopolymer, with interesting properties and the second most abundant biological polysaccharide in the world [22] is obtained by chitin deacetylation, which is present in the crustaceans exoskeletons, insects, mollusks and fungal cell wall [23, 24].

Currently, fibers have been successfully generated by electrospinning of chitosan solutions in 90 wt.% aqueous acetic acid [25, 26] or using trifluoroacetic acid (TFA), that can be associated with dichloromethane (DCM) [27].

Recent studies have indicated the importance of structural–dynamic characterization of ultrathin chitosan fibers [28] and pointed out their wide range of applications, as oral insulin stability [29], guided bone regeneration [30-32], grafts for alveolar bone defects and microbicides nanocomplexes for protection against HIV infections [33].

In addition to this, many studies have associated chitosan with other components such as silver nanoparticles [34, 35], collagen [36, 37] and natural extracts [38] to electrospun fibers with a wide variety of properties, such as antibacterial activity [34, 35], alternative wound healing [38] and for suitable cell proliferation environments [36, 37].

In this context, hydroxyapatite, an important component of bone mineral phase, is a nanomaterial that can be incorporated in chitosan fibers for bone regeneration activity [39]. Furthermore, at the nanoscale, nanohydroxyapatite (nHAp) form has high applicability from anti-erosion dentifrices [40] until tissue engineering [41, 42].

This particle-size was produced, in this study, to be incorporate in chitosan fibers because of their similarity to bone tissue hydroxyapatite by reducing biomimetic limitations.

Accordingly, the current study proposed to synthesize and characterize electrospun chitosan/nHAp fibers, in order to evaluate the nHAp morphological, physicochemical and biological influences, under specific electrospinning parameters combinations.

2. Methods

2.1 Materials and Experimental procedure

In this study, nHAp crystals were prepared via aqueous precipitation method [43]. Briefly, the nHAp particles were produced using: 10 mL of 0.167M of $\text{Ca}(\text{NO}_3)_2 \cdot 4\text{H}_2\text{O}$ and 100mL of 0.1M of $(\text{NH}_4)_2\text{HPO}_4$. NH_4OH was added in order to keep the pH~10. The precipitate was subjected to ultrasonic irradiation for 30 min (Ultrasonic Processor 500 W;20 kHz; 13 mm probe; model: SO-VCX-500, SONICS). The resulting suspensions were left to age for

120h. Subsequently, they were filtered, washed with water and dried at 60°C for 4h. This procedure is aimed when a Ca/P of 167 is desired [44].

A typical chitosan solution 7 % (w/v) was prepared by dissolving 0.7 g of medium molecular weight chitosan (75-85 % deacetylated, Sigma-Aldrich, Saint Louis, USA) in 7 mL of Trifluoroacetic acid (TFA) (99% purity, Sigma-Aldrich, Saint Louis, USA) and 3 mL of Dichloromethane (DCM) (98% purity, Sigma-Aldrich, Saint Louis, USA), under constant stirring.

For the chitosan/nHAp solutions, were dissolved 0.7 g of chitosan in 7 mL of TFA, under constant stirring at 30°C. Next, 0.05 g of nHAp were dispersed in 3 mL of DCM and added to the chitosan/TFA solution.

The electrospinning was performed after stirring: solutions were placed in 5 mL plastic syringes (BD, Juiz de Fora, MG, Brazil) coupled to a metallic needle (0.7 mm, INJEX, São Paulo, SP, Brazil) and subjected to the electrospinning process, under different combinations of three parameters: distance (10 to 15cm), voltage (10 to 20kV) and flow rate (0.1 to 0.5 mLh⁻¹).

The fibers were collected at room temperature (25°C) and relative humidity of 30 %, in a collector (10x15 cm) covered with aluminum foil.

The membranes were immersed in 5 M Na₂CO₃ aqueous solution for 3h, at room temperature, in order to neutralize the TFA salts [45]. After the immersion, the membranes were washed with distilled water and dried at room temperature for 24h.

2.2 Morphological and topographical characterization

Scanning Electron Microscope (Inspect S 50, FEI Company, Brno, República Tcheca) (under high vacuum, 15-25 kV and spot 5.0) was used to investigate the fiber's morphology. To taken micrographs a thin layer of gold

under low atmospheric pressure (SC7620 'Mini' Sputter Coater/Glow Discharge System, Emitech, East Sussex, UK) were evaporated on the mats.

*The mean diameter of the fibers was calculated using ImageJ software (Versão 1.44o, National Public of Health) from SEM micrographs. Therefore, four samples were selected, two Ch and two ChHa, each one originated from a specific combination of synthesis parameters. **Table 1** shows the groups evaluated and individually parameters to collect micrographs.*

The topographic roughness parameters (R_a = arithmetic average) of membranes were measured by an optical profilometer (Wyko NT 1100, Veeco, Plainview, NY, EUA).

For this study, we used an Atomic Force Microscope (AFM) (Veeco Multimode V, Veeco Instruments, Plainview, NY, USA) located at the LAS/INPE, under intermittent contact mode.

2.3 Physico-chemical characterization

FTIR (Spotlight 400, Perkin-Elmer, Waltham, MA, USA), with an accessory of attenuated total reflectance (ATR-FTIR), was performed at the Instituto de Pesquisa e Desenvolvimento (IP&D-Univap), from 4000 to 500 cm^{-1} , using 32 scans and a 4 cm^{-1} resolution.

Thermogravimetry was performed on samples (6-8 mg) placed in a platinum pan and heated from 25°C to 900°C (10°C min^{-1}), under N_2 atmosphere (flow 50 mL^{-1}) (TGA-50TA, Shimadzu).

2.4 Biological tests

All animal procedures are performed in accordance with the ethics committee guidelines of Sao Jose dos Campos Institute of Science and

Technology, School of Dentistry – UNESP (006/2016-CEUA-ICT-UNESP). And it were realized in accordance with the Ethical Principles for Animal Experimentation, adopted by the Colégio Brasileiro de Experimentação Animal (CONCEA).

After obtained from femurs young adult male Wistar rats, Mesenchymal Stem Cells (MSCs) were cultured in α -MEM (Gibco, Life Technologies, Grand Island, NY, USA), supplemented with 10 % fetal bovine serum (Gibco), 50 mg/ml gentamicin (Gibco), 5 mg/l ascorbic acid (Gibco) and 2.16 g/l α -glycerophosphate (Sigma). Afterwards, the cells were cultured in 24 microplate wells and incubated at 37 with 5 % CO₂. [46]

As control, cells were grown onto culture plate (TCP), and the other groups assessment were Chitosan samples (Ch) and chitosan/nHAp samples (ChHa), for all in vitro analysis.

After 7 days for cultured, MTT colorimetric assay were performed with incubation in 3-(4,5-dimethylthiazol-2-yl)-2,5-diphenyltetrazolium bromide (M5655 Sigma-Aldrich St Louis MO, USA), then the cells were lysed with a propanol acid solution (2-Propanol Sigma-Aldrich St Louis MO, USA) for colorimetric measurement in a spectrophotometer (570 nm with correction at 650 nm) (EL 808 BioTek Instruments, Winooski, USA), as previously describe [47]. The values were expressed in percentage of viable cells compared to the control group.

The data of alkaline phosphatase activity were obtained in quintuplicate, by the release of thymolphthalein by hydrolysis of the thymolphthalein monophosphate substrate, using a commercially available kit (Labtest Diagnóstica SA — Ref. 40, Lagoa Santa, Minas Gerais, Brazil) following the manufacturer's instructions. The test was performed as previously described by Prado et al. [47]. Absorbance was measured in a spectrophotometer at 590 nm and the alkaline phosphatase activity was normalized by total protein content,

which was determined by Lowry method [48], and the data were expressed as $\mu\text{mol thymolphthalein/h/mg protein}$.

3. Results and discussion

3.1 Morphological and topographical characterization

The electrospun fibers presents high applicability in tissue regeneration [45-47] by controlling their morphology, porosity and flexibility, being effective to cell scaffolds [48] and topographic surface can change cytoskeletal arrangement and focal adhesion formation [49].

Figure 1 shows Scanning Electron Micrographs (SEM) of electrospun fibers. A-Ch shows the typical structure in the electrospun chitosan for 0.3 mL/h, 12 cm, 10 kV and B-Ch for 0.3 mL/h, 15 cm, 10 Kv. C-ChHa shows the electrospun chitosan/nHAp for 0.5 mL/h, 12 cm, 17 kV and D-ChHa for 0.5 mL/h, 15 cm, 17 kV. All Ch samples showed homogeneous fibers' networks, whilst all samples obtained from ChHa presented beads and some discontinuities, as can be seen in.

Some studies demonstrated that higher voltage offers the greater probability of beads formation [49, 50] and voltage can influence the fiber diameter, but should vary according to the polymer solution and on the distance between the capillary and the collector [51].

In this study, temperature ($25^{\circ}\text{C} \pm 2$) and relative humidity ($40\% \pm 5$) are standardized whereas these elements were extremely important to the electrospinning effectiveness [50-52]. There are reports that relative humidity can modify the fibers diameter [52] and temperature variation can also cause this change, which can be associated with the solvent evaporation rate and solution viscosity [52]. As expected, Ch and ChHa solutions were working

parameters on electrospinning sensitive and the effective fiber formation was only found under specific combinations (**Table 1**).

The neutralization procedure was necessary to realize the scaffolds usefulness in areas that require a contact with neutral or weak basic aqueous middle [45] and the environment neutralization should be equally considered to obtain optimal water absorption [53].

The average diameter fiber ranged from 324 nm (D-ChHa) to 763nm (B-Ch). It was also observed sensibility to the nHAp loading, with a significant ChHa diameter decrease. In the **Fig 2** the measured values distribution is presented in histograms.

The diameter distribution presented significant difference between A-Ch and B-Ch, that could be related with distance decrease, between the capillary and the collector, which culminates in the increase of electric field, responsible for fibers stretching, implying a diameter fiber decrease [54, 55].

During an electrospinning process, it is very important to produce the fibers with minimum possible diameter, due the thinner and homogeneous fibers that provide maximum surface area and porosity, which results in a higher adsorbents efficiency [56] Thus, it is aimed to obtain a biomimetic characterization of the extracellular matrix.

It was observed that fibers formed at lower collector distance (7.5 cm), presented bead formation. It could be due to the higher electrical forces on the jet which made it unstable. Under controlled distance increase (13.75 cm), the solution jet stability was also increased, in order to a homogeneous fibers synthesis and lower diameters. In addition to that, with 20 cm distance, fibers with higher diameter were formed, perhaps, accordingly to the strength of electrical force on the spinning solution decrease [57].

In this sense, is reported in literature that the diameter could be influenced by whether complete crystallization of polymer chains took place before or after the electrospinning jet has reached the collector [58].

*The scaffold surface morphology has an important role in tissue engineering [59], and studies have shown the chemistry, geometry and topography importance to cell behavior [60, 61]. **Figure 3** shows the images obtained from A-Ch, B-Ch, C-ChHa and D-ChHa where it is possible to analyze 2D (right) and 3D (left) perspectives.*

This analysis presented more homogeneous Ch than ChHa fibers. There was average Ra (Roughness average) $9.22 \mu\text{m} \pm 1.94$ for Ch and $2.92 \mu\text{m} \pm 0.29$ for ChHa, with statistically significant difference between them ($p = 0.0052$). Thus, the ChHa diameter fiber reduction can be attributed to the fact that electrospun polymer nanofibers, with the lowest fiber diameter, were obtained at the highest electrical conductivity [62] and consequent increase the neat charge density in the jet due to nHAp presence.

*In this context, AFM analysis (**Fig 4**) showed that amounts of nHAp altered the membranes surface morphology.*

*Through the AFM analysis it was possible to observe better homogeneity and smoothness in the Ch fibers compared to the high roughness of ChHa. **Figure 4.A** showed a more homogeneous surface and clear disclosure of ultrathin fibers with about approximately $0.6 \mu\text{m}$. In contrast, **Fig 4.B** showed an irregular and rough surface. It can be explained by the nHAp increased content, that changes scaffolds microstructure, attributed to the gradual weakening of the intermolecular interactions as the hydrogen bonding between the chitosan fiber molecules [63].*

*The FTIR spectra of all samples are shown in **Fig 5.A**.*

*It is possible to identify the characteristic bands of electrospun Ch and ChHa (**Fig 5.B**) at 720 , 800 and 840 cm^{-1} . The nHAp analysis shows a band in*

the range of 550 cm^{-1} . Already at 1644 cm^{-1} and 1552 cm^{-1} transmittance occurs, as seen in **Fig 5.C**.

The FTIR analysis presented characteristic transmittance bands that reveals TFA salt formation (**Fig 5.B**) settling to three bands (720 , 800 and 840 cm^{-1}) - that disappeared after alkaline treatment - and amino group ($-\text{NH}_3^+$) around 1675 and 1540 cm^{-1} [64].

In fact, the chitosan amino groups are reported as *D*-glucosamine residues which can be protonated, providing solubility in acid aqueous solutions [65, 66].

Chitosan dissolved in TFA is a very strong acid dissolution that can occurs not only by a protonation of the amino groups but also by esterification, and this esterified product is more soluble than chitosan by itself, and which, by FTIR verified the presence of TFA salts, that can explain the immediate dissolution of ChHa and Ch scaffolds in water presence.

For this reason, it is understood that the TFA choice, as chitosan solvent, must aim the neutralization and residual elimination [45, 67] on the electrospun ultrathin fibers, as an important applicability feature in the cellular proliferation context.

The same amine groups are reported in the salt formation when combined with TFA. It was observed that the TFA destroyed the intermolecular interactions between the chains of chitosan and facilitated the electrospinning process [68]. This is showed in transmittance bands by the FTIR analysis, that according to some studies disappear after alkaline treatment [67].

Regarding to nHAp, OH flexural vibration appeared as a weak peak in the low-frequency band of 550 cm^{-1} as some studies [69] found in 669 cm^{-1} , that verifies the presence of amino groups ($-\text{NH}^3$). Simultaneously, 1644 cm^{-1} and 1552 cm^{-1} absorbance can be attributed to the two types of variations, H-O-H (water) flexural vibration and carbonyl stretching ($\text{C}=\text{O}$), respectively [64, 70].

Studies it was observed that observed OH stretching (3500 cm^{-1} FTIR) when nHAp was subjected to high temperatures ($> 800\text{ }^{\circ}\text{C}$) [71]. Moreover, at high temperatures ($> 600^{\circ}\text{C}$), this weight loss may be assigned to decarboxylation of nHAp, releasing CO_2 (g) and causing condensation of HPO_4^{2-} releasing water [71].

Thus, chitosan hydrophilicity is partly influenced by the presence of amine groups, that are vital in the dissolution context of this material in an acid solution and in the resistance of chemical and biological degradation [72].

According to thermal tests, the results were included at different stages (Table 2). The first one correspond to absorbed/adsorbed loss of water, and the second one, was about thermal decomposition, until reaching a maximum degradation, under a maximum decomposition temperature.

The electrospun Ch had the first mass loss stage at $25\text{ }^{\circ}\text{C}$ to $118\text{ }^{\circ}\text{C}$ range, with a mass loss of 9.0% ; a second stage in the range $126\text{ }^{\circ}\text{C}$ to $280\text{ }^{\circ}\text{C}$, with a weight loss of 40.1% ; and a third and final stage, between $280\text{-}431\text{ }^{\circ}\text{C}$, with a weight loss of 24.0% . Figure 6 shows the thermogravimetric analyses of a chitosan sample average molecular weight of powder, chitosan and electrospun chitosan with nHAp.

ChHa had the first stage between $25\text{ }^{\circ}\text{C}$ and $120\text{ }^{\circ}\text{C}$, with a weight loss of 10.9% ; the second stage between $125\text{ }^{\circ}\text{C}$ and $288\text{ }^{\circ}\text{C}$, with a weight loss of 40.1% ; and a third and final stage, between $288\text{-}435\text{ }^{\circ}\text{C}$, with a weight loss of 21.5% .

Already isolated nHAp features three weight loss stages, in the ranges of $25\text{-}120\text{ }^{\circ}\text{C}$ (loss of 4.58%), $120\text{-}390\text{ }^{\circ}\text{C}$ (loss of 5.78%) and $390\text{-}500\text{ }^{\circ}\text{C}$ (loss of 1.1%), all relating to chemical and physical loss of water. Among $500\text{-}1000\text{ }^{\circ}\text{C}$, the mass nHAp varied very little (loss of 0.11%).

For the biological tests, the cell viability was obtained by MTT test, and the results showed in Fig 7, measurement of cell viability (viable cells

comparing to control group), that the biomaterial influenced positively the cell culture. Therefore, there was no statistical difference between Ch and ChHa ($p > 0.001$). The control group exhibited a lower quantify of cell proliferation showing statistical difference with experimental groups ($p < 0.001$). The results showed that the scaffolds did not present cytotoxicity effects once the cell proliferation it bigger than control group cytotoxicity effects.

The cell-material interaction include adhesion and spreading of osteoblast cell on the material surface, which will influence following processes of cell proliferation and differentiation [73]. Previous studies observed that the material surface properties can regulate these process [74, 75], with roughness being important factor that affects cell adhesion and proliferation [76]. Although the roughness rate between the experimental groups showed a statistical difference, this factor did not influence the cell viability rate. However, a tendency was observed for the viable cell increase in the rougher fibers represented by ChHa.

It is suggested, from the present study, that this value was similar between the experimental groups because the Ch fiber exhibited another factor, that acts on the cellular behavior, a structural arrangement, allowing also favorable cell viability. Then, both fibers constituted the propitious cell metabolism environment.

The rates of cell viability exhibited by both groups make them suitable for biomedical use according to the International Organization for Standardization (ISO) and national standards for Biological Evaluation of Medical Devices, ISO 10993-5 [77].

Regarding to osteoblastic differentiation, the association of hydroxiapatite with polymers have present positive results into cell metabolism [78]. Statistical difference was presented among the groups ($p < 0.001$), the ChHa group provided higher means of ALP activity (**Fig 8**), spite without

different with Ch group. Alkaline phosphatase is known to be an early indicator of osteoblast differentiation, and your expression is considered to be a functional parameter in vitro, reflecting the progress of cell differentiation [79].

It is important to note that among all natural polymers, chitosan (CS) was selected for this study due to its unique structural characteristics similar to GAGs in the ECM [76], and it is suggested that its association with nHAp did not provide greater cell differentiation capacity due to the great positive chitosan influence.

The results exhibited represents that these new biomaterials are acceptable in the biological environment, with the osteogenic differentiation possibility, and can provide evidence for the potential of biomimetic ultrathin chitosan fibers in tissue engineering and clinical application [80].

Thus, it was reasonable to conclude that the prepared nanofibrous matrix coated with chitosan and nHAp could be potential candidates for the regeneration of bone, once that the tendency can be observed in higher level of ChHa group cell activity.

4. Conclusions

Thus, it is concluded that the addition of nHAp can influence the topographical and morphological aspects of ultrathin chitosan fibers, in order to reduce the average diameter and to increase surface roughness, which can change its behavior towards by the modification in the development and cell proliferation.

In contrast, nHAp crystals did not influence the physicochemical characteristics of the chitosan scaffolds.

And the fibers biological tests with and without nHAp showed to be a good environment for cell proliferation and differentiation, which open a wide

range of applications of these materials for tissue engineering and clinical potentials.

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LIST OF TABLES

Table 1 - *Electrospun samples obtained under different electrospinning parameters: A-Ch, B-Ch, C-ChHa e D-ChHa*

Solution	Parameters	Specimen
(Ch)	0.3 mLh^{-1} -12 cm-10 kV(A)	A-Ch
(Ch)	0.3 mLh^{-1} -15 cm-10 kV(B)	B-Ch
(ChHa)	0.5 mLh^{-1} -12 cm-17 kV(C)	C-ChHa
ChHa)	0.5 mLh^{-1} -15 cm-17 kV(D)	D-ChHa

Table 2 - Weight loss stages associated with the evaporation of water absorbed / adsorbed and decomposition obtained by TG and dTG

Sample	1° Stage (water loss)	2° Stage (decomposition)	3° Stage (decomposition)
Chitosan	25-168 °C →5.04 %	202-433 °C →35.99 %	-
Electrospun Ch	25-118 °C →9.04 %	126-280 °C →40.06 %	280-431 °C →24.03 %
Electrospun ChHa	25-120 °C →10.90 %	125-288 °C →40.15 %	288-435 °C →21.47 %

FIGURES

Figure 1 - SEM images of the electrospun products from A-Ch, B-Ch, C-ChHa and D-ChHa

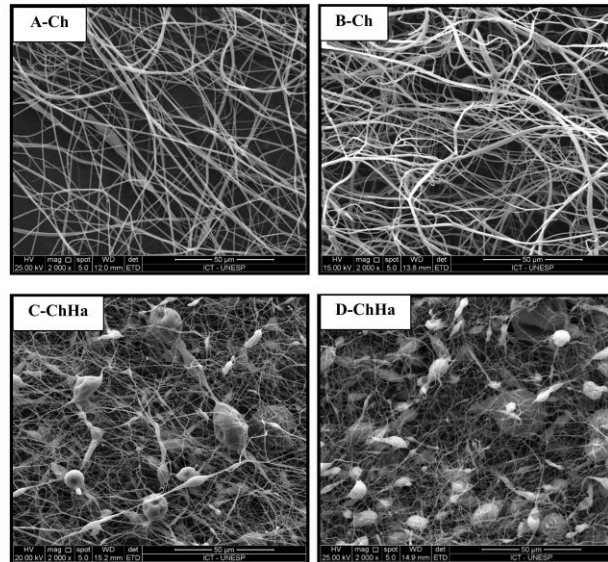


Figure 2 - Average fibers diameter histograms: A-Ch, B-Ch, C-ChHa and D-ChHa.

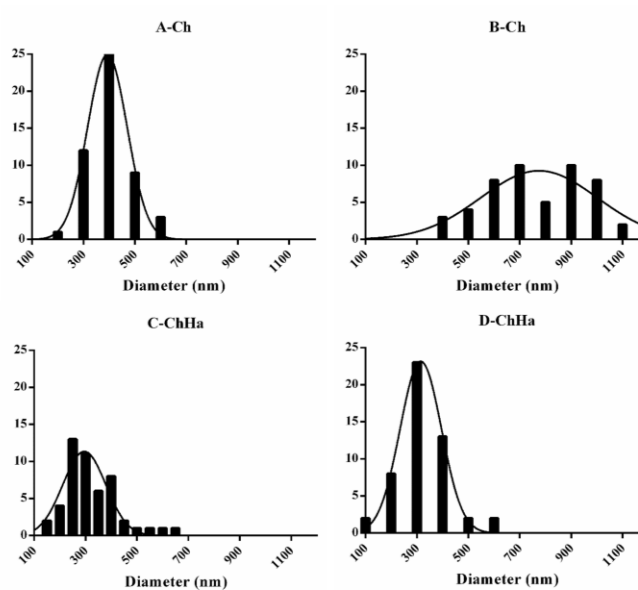


Figure 3 - A-Ch, B-Ch, C-ChHa and D-ChHa profilometry analysis in 2D and 3D images

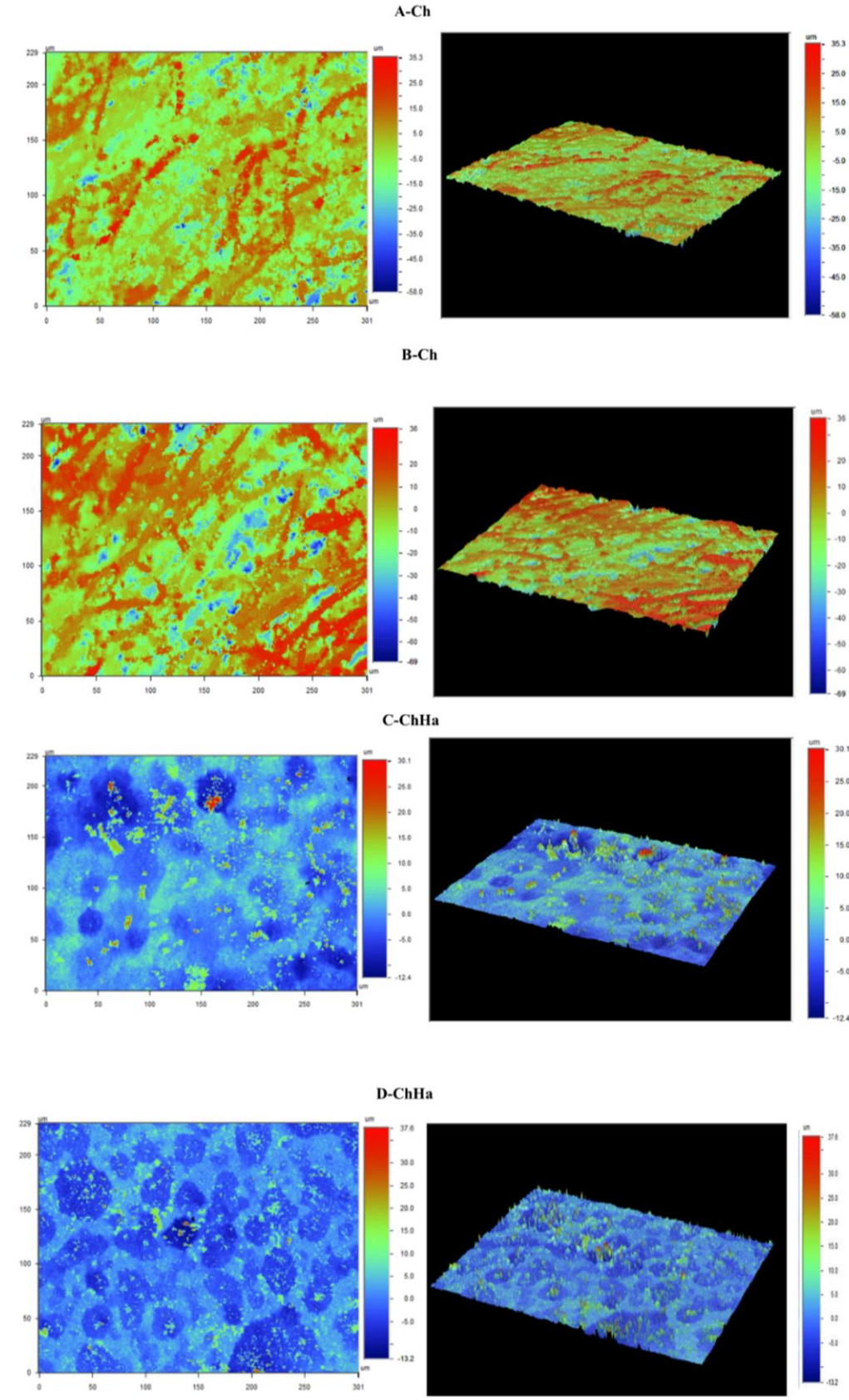


Figure 4 – AFM analysis A: Electrospun Ch AFM analysis and B: Electrospun ChHa

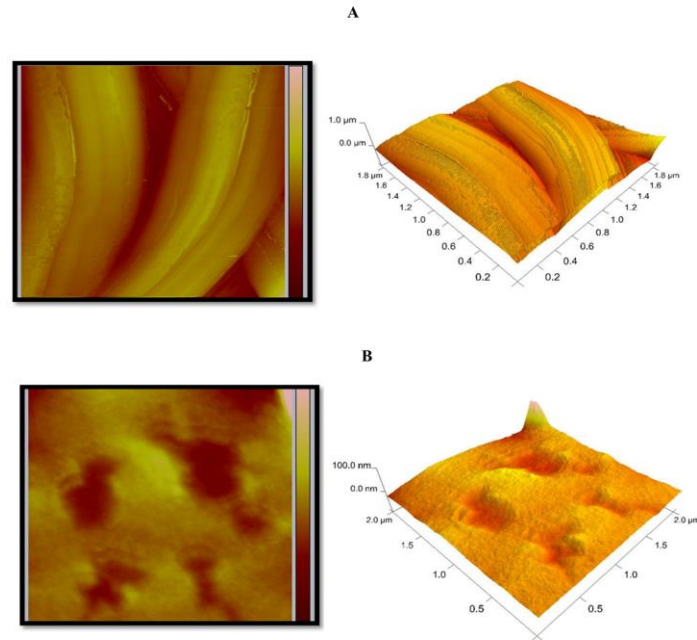


Figure 5 – A) FTIR analysis of all samples B) Characteristic bands from Ch and ChHa and C) Characteristic bands from nHAp

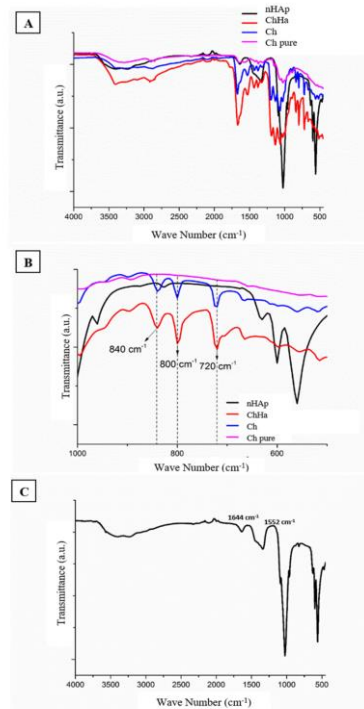


Figure 6 - (a) TG curves and (b) dTGs samples of Ch, electrospun Ch, electrospun ChHa and nHAp.

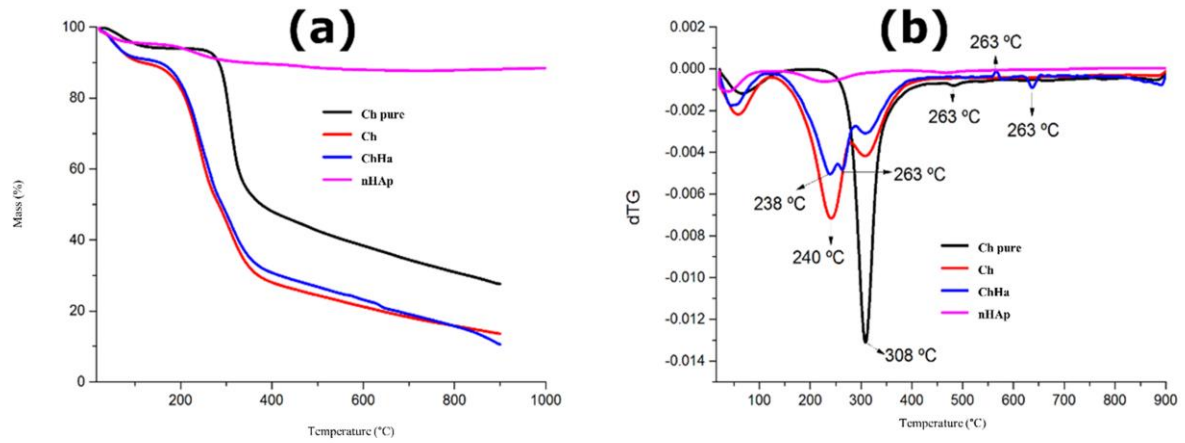


Figure 7 - Measurement of cell viability MTT assay

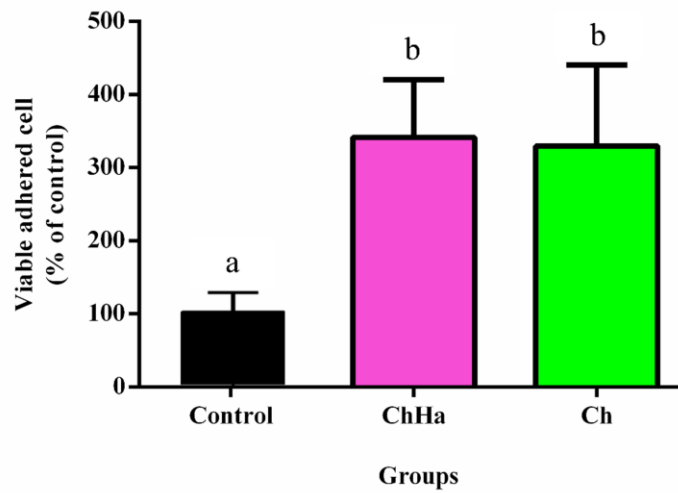
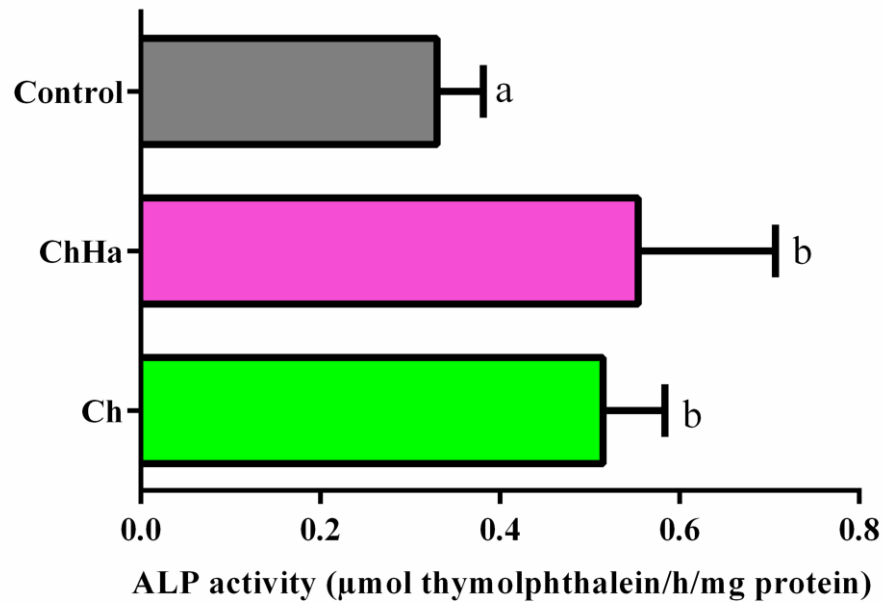
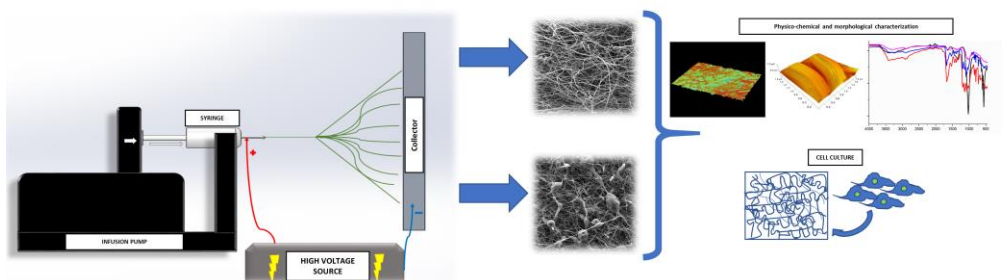


Figure 8 – Osteogenic differentiation by Alkaline phosphatase activity of MSCs cells after 10 days of incubation on control group, Chitosan samples (Ch) and chitosan / nHAp samples (ChHa). Values are reported as mean \pm SD ($n = 5$).



Graphical Abstract



2.2 Artigo – Sato TP, Mello DCR, Vasconcellos LMR, Valente AJM, Borges ALS. Desenvolvimento de matrizes de hidrogel de quitosana-DNA e quitosana-pectina para sistema de entrega de fitoterápicos / *Development of Chitosan-DNA and Chitosan-Pectin hydrogel matrices for herbal medicine delivery system**

RESUMO

Este estudo teve como objetivo sintetizar hidrogéis de quitosana-DNA (CS-DNA) e quitosana-pectina (CS-P) para administração de fitoterápicos. Para isso, os hidrogéis foram preparados pela técnica de emulsão (óleo em água): as emulsões foram obtidas pela mistura da fase aquosa da solução de CS (1 mL) e solução de DNA (1 mL) (1:1 razão de quitosana:DNA) ou Solução de P (1 mL) (1:1 razão de quitosana:pectina) com álcool benzílico (5 mL) usando um instrumento de dispersão de alto desempenho. O Própolis Verde (PV) foi incorporado por embebição: 20 mg hidrogel numa solução aquosa de PV (70 µg/mL) por 2h e a cinética de liberação do PV foi analisada a 25 e 37 °C em água e saliva artificial. Os espécimes obtidos foram liofilizados e depois caracterizados utilizando diferentes técnicas e a viabilidade celular *in vitro* foi realizada utilizando a linha celular MG63. A presença de pectina foi evidenciada pela ocorrência de uma forte banda a 1745 cm⁻¹, também ocorrendo na mistura, atribuída aos grupos esterificados. As bandas que ocorrem em 1105 e 1010 cm⁻¹, na região das impressões digitais, são típicas dos polímeros de pectina e podem ser atribuídas ao alongamento C=O. Nos géis de DNA e CS-DNA, é possível observar uma forte banda a 1650 cm⁻¹, atribuída à vibração de estiramento C2=O2 da citosina. A sorção de PV induziu uma modificação significativa da morfologia da superfície do gel. A partir do TGA, pode-se concluir que ocorre alguma separação de fases entre a quitosana e o DNA. A eficiência do encapsulamento não muda significativamente com a temperatura. A cinética de liberação na água ou na saliva segue um mecanismo de duas etapas. E os resultados biológicos exibiram que esses materiais são aceitáveis no ambiente biológico e podem fornecer evidências para o potencial de material biomimético. Pode-se concluir que a complexação de hidrogel de CS com DNA e pectina é uma alternativa viável ao sistema de administração de medicamentos à base de plantas como o PV.

Palavras-chave: Hidrogel. Quitosana. Pectina. DNA. Própolis.

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ABSTRACT

This study aimed to synthesize Chitosan-DNA (CS-DNA) and Chitosan-Pectin (CS-P) hydrogels for drug delivery of herbal medicine. For this, hydrogels were prepared by emulsion technique (oil-in-water): emulsions were obtained by mixing aqueous phase of the Ch solution (1 mL) and DNA solution (1 mL) (1:1 ratio of Chitosan:DNA) or Pectin solution (1 mL) (1:1 ratio of Chitosan:Pectin) with benzyl alcohol (5 mL) using a high-performance dispersing instrument. Green Propolis (GP) was incorporated by imbibition: 20mg of hydrogel in 70 µg/mL GP aqueous solution for 2h and GP release kinetics was analyzed at 25 and 37°C in water and simulated saliva. The obtained specimens were freeze-dried and then characterized using different techniques and in vitro cell viability was performed using MG63 cell line. The presence of pectin was evidenced by the occurrence of a strong band at 1745 cm⁻¹, also occurring in the blend, ascribed to esterified groups. The bands occurring at 1105 and 1010 cm⁻¹, in the finger-print region are typical of pectin polymers and can be attributed to the C=O stretching. In the DNA and chitosan-DNA gels it is possible to observe a strong band at 1650 cm⁻¹, slightly shifted from the chitosan band, assigned to the C2=O2 stretching vibration of cytosine. The sorption of GP induced a significant modification of the surface morphology of the gel. From TGA it can be concluded that some phase separation occurs between chitosan and DNA. The encapsulation efficiency does not significantly change with temperature. The release kinetics either in water or saliva follow a two-step mechanism. And the biological results exhibited that these materials are acceptable in the biological environment and can provide evidence for the potential of biomimetic material. It can be concluded that the hydrogel complexation of CS with DNA and Pectin is a viable alternative to drug delivery system of herbal medicine such as GP.

Keywords: Hydrogel. Chitosan. Pectins. DNA. Propolis.

INTRODUCTION

Hydrogels are known as crosslinked polymeric 3D networks capable of absorbing and retaining large amounts of water or biological fluids^{1 2}, having a microstructure similar to the extracellular matrix (ECM)² and a biocompatibility to blood and biological tissues³. Besides, hydrogels can be

designed with responsive properties which make them highly appreciated for biomedical and pharmaceutical (drug/gene deliver) applications ⁴⁻⁷.

Thus, much effort has been made on the hydrogel-based matrices for the development of vectors for the delivery of active substances for different targets. Among different strategies, the use of biopolymers is mandatory. Following such approach, chitosan (Chi) is a cationic biopolymer, obtained from the chitin desacetylation of crustaceous exoskeleton ⁸, *showing antimicrobial biodegradable and biocompatible properties. Furthermore, due to the presence on its structure of amine and hydroxyl groups* ^{9, 10}, *the Chit is highly chemically versatile, being used for several applications in pharmaceutical and medical areas* ¹¹, *especially for wound healing applications* ¹² *as they are similar to the natural extracellular matrix (ECM).*

The dissolution of chitosan occurs in slightly acidic (pH < 6.5) aqueous solutions, as a consequence of the protonation of its amine groups, allowing the formation of either chemical or physical crosslinked hydrogels ¹³. *Covalently crosslinked chitosan hydrogels are very effective in drug delivery systems* ^{14, 15}; *however, they may show some drawbacks as, for example, the use of toxic crosslinkers or initiators. The use of physical hydrogels allow to overcome such limitation* ¹⁶. *Among different approaches, and taking advantage of the positively charged chitosan, we have decided to investigate the formation of chitosan-DNA and chitosan-pectin coacervate hydrogels. Deoxyribonucleic acid (DNA) is a biopolymer composed of deoxyribose sugar, able to carry genetic information* ¹⁷. *DNA hydrogels have unique properties such as biocompatibility, selective binding, and molecular recognition* ^{18, 19}, *creating a wide range of potential applications in drug delivery* ^{6, 20} *and tissue engineering* ²¹⁻²³. *In fact, among biological polyelectrolytes, DNA has always attracted particular interest, and there are numerous studies of the interactions between DNA and polycations* ²⁴⁻²⁷. *The formation of coacervate hydrogels is achievable once DNA*

is completely ionized at neutral pH²⁸ enabling the interaction with positively charged molecules, such as chitosan²⁹.

Pectin (P) is another polyanion at pH higher than ca. 9 that can form coacervate hydrogels with chitosan^{30, 31}. Pectin is a plant cell wall polysaccharide used as a gelling and stabilizing polymer in food or biomedical products due the influence on human health such as cholesterol levels and local bleeding^{32, 33}.

These two matrices were tested as a sustained drug release matrix for Green Propolis (GP). This is an herbal medicine produced by *Apis mellifera* honeybees, from *Baccharis dracunculifolia* DC (Asteraceae), and used as anti-inflammatory³⁴, antifungal³⁵ and antioxidant^{36, 37} drug, due to its high levels of phenolic acids³⁸.

MATERIALS AND METHODS

Materials

Chitosan with acetylation degree of 15 mol% (M_w : 87×10^3 g mol⁻¹) was purchased from Golden-Shell Biochemical (China). Pectin from citrus peel and double-stranded (ds) DNA (sodium salt from salmon testes), with a polymerization average degree of ca. 2000 base pairs, were purchased from Sigma. Benzyl alcohol (99%) was obtained from Merck KGaA (Germany).

Aqueous extract of Green Propolis was obtained from Apis Flora (Ribeirão Preto, SP, Brazil). Millipore-Q water has been used to prepare the solutions.

Simulated saliva was made by Indiana University modified method³⁹, prepared according to Fusayama et al (1963)⁴⁰.

Hydrogels preparation

Both cocervate hydrogels (chitosan-DNA and chitosan-pectin) have been prepared by emulsion technique (oil-in-water). The preparation of the former gel has been done using an acetate buffer (pH 6). This buffer was chosen to balance the cationic density of chitosan and the chemical instability of DNA ⁴¹.

Thus, a chitosan aqueous solution (1 % w/v) was initially prepared in acetate buffer and then filtered through a paper filter to remove insoluble substances ³¹. *DNA and pectin aqueous solutions were prepared in phosphate buffer at pH 6 and pH 9.2, respectively, under stirring for 12 hours, at room temperature. The emulsions were obtained by mixing aqueous phase of the Ch solution (1 mL) and DNA solution (1 mL) (1:1 (v/v) ratio of Ch:DNA) or Pectin solution (1 mL) (1:1 (v/v) ratio of Ch:Pectin) with benzyl alcohol (5 mL) using a Ultra-Turrax at 31-34000 rpm min⁻¹ for 5 minutes* ⁴².

Hydrogel Characterization

The synthesized hydrogels were freeze-dried previously to their characterization by different techniques. Attenuated reflection infrared spectroscopy (ATR-FTIR) was carried out in a Varian Cary 630 FTIR Spectrometer, with wavenumber ranging from 650 to 4000 cm⁻¹.

The thermogravimetric analysis (TGA) were performed in a thermogravimetric analyzer (TG209 F3 Tarsus Netzsch Instruments). Samples of ca. 5 mg were weighed in alumina pans and heated from 25 °C to 800 °C at a 10 °C min⁻¹ rate, under N₂ atmosphere (20 mL min⁻¹).

The surface morphologies of hydrogels have been analyzed by scanning electron microscopy (SEM) with a JEOL model 5310 scanning microscope operating under low vacuum at 15 kV. The membranes were submitted to a fast cryogenic treatment by diving gel samples into liquid nitrogen for 10 s, and then they were left overnight in a freeze dryer (Free Zone 4.5-Labconco) before being coated with a gold film.

The effect of water and simulated fluids on hydrogels before and after the loading of Green Propolis was assessed by swelling degree (Q). Different samples of the same membrane were cut, weighed, and immersed in water or simulated solutions and left to reach the swelling equilibrium (2days). After this time, membranes were removed from the solution, any drops of solution were wiped off, then the weight of the membrane was measured using an analytical balance (ADA 120LE, (0.1 mg). Each experiment was repeated twice. The swelling degree of the gel (Q) was calculated by using the equation

$$Q = \frac{m_w}{m_x} \quad (1)$$

where m_w and m_x are the masses of water (or solution) in the swollen gel and in the xerogel, respectively. The mass of xerogel samples has been obtained using the weight of the sample after synthesis and taking into account the solid content of the gel⁴³.

Loading and release of Green Propolis

Green Propolis has been incorporated into coacervate gels by imbibition. A 70 μ g/mL GP aqueous solution was prepared; after that, hydrogel samples (of ca. 20 mg) were placed in that volume of GP solution and left there for 2 hours, at 25 °C, using a thermostatic bath.

The encapsulation (EE) efficiency was calculated by using, respectively, the following equations:

$$EE = \frac{(\text{total amount of GP} - \text{non-bound GP})}{\text{total amount of GP}} \times 100 \quad (2)$$

To determine the GP release kinetics the hydrogels samples, after GP encapsulation, were placed in vessels containing 20 mL of water, and kept in an incubator (Labwit, ZWY-100H), at 37 °C and 150 rpm. At predetermined time intervals (20 min), an aliquot of the supernatant were taken from the vessel and

storage for GP quantification. Meanwhile, that volume was replaced by water. The process was repeated until the release equilibrium is attained. The GP released into the supernatant was quantified spectrophotometrically by measuring the absorbance at range of 200 to 800nm using a double-beam Shimadzu UV-2100 spectrometer (SHI-MADZU EUROPA, Germany).

Biological Analysis

In this present study were used established lineage of osteoblast-like cells (MG63) acquired from Rio de Janeiro Cell Bank (Rio de Janeiro, RJ, Brazil). Cells were cultured in Modified Eagle's medium (Life Technologies, NY, USA) containing 10 % fetal bovine serum (Life Technologies, NY, USA) and penicillin (100U/mL) at 37 C in a humidified atmosphere of 5 % CO₂. The culture medium was changed to each two days until moment test. All tests were performed according to ISO 10993 guidelines, with three independent experiments.

Initially, five samples for each material were placed into 24 well plates (TPP, Curitiba, Brazil), and followed 2×10^4 cells were placed in each well. After 7 days for cultured, MTT colorimetric assay were performed with incubation in 3-(4,5-dimethylthiazol-2-yl)-2,5-diphenyltetrazolium bromide (Sigma-Aldrich St Louis MO, USA), then the cells were lysed with a propanol acid solution (Sigma-Aldrich St Louis MO, USA) for colorimetric measurement in a spectrophotometer (EL 808 BioTek Instruments, Winooski, USA) at 570 nm, as previously describe ⁴⁴. The values were expressed in percentage of viable cells compared to the control group (only cells) considered 100%.

Proteins were extracted with 0.1 % sodium dodecyl sulfate (Sigma-Aldrich St Louis MO, USA), followed by addition of Lowry solution (Sigma-Aldrich St Louis MO, USA). The extract was diluted in Folin–Ciocalteu reagent (Sigma-Aldrich St Louis MO, USA) and the absorbance was assessed in a spectrophotometer (8582 Micronal, São Paulo, Brazil) at 680 nm ⁴⁴. The total

protein content was calculated from a bovine serum albumin standard and expressed in $\mu\text{g/mL}$.

RESULTS AND DISCUSSION

Gels characterization

Figures 1 and 2 show the FTIR spectra of the chitosan-pectin and chitosan-DNA blend hydrogels, respectively, and the corresponding isolated polymers. The IR spectra of chitosan (Figures 1 and 2) show two characteristic bands at 1648 and 1538 cm^{-1} assigned to (C=O) stretching mode and N-H bending of amide group. A band at 1378 cm^{-1} can be attributed to the $-\text{CH}_3$ symmetrical deformation and the strong band at 1076 cm^{-1} characterizes the C-O stretching^{45, 46}. Furthermore, the spectra show bands at 3417 cm^{-1} that indicates dimeric O-H stretch and at 1425 cm^{-1} indicating aromatic C-C stretch⁴⁷.

To confirm the occurrence of the coacervation process, the presence of pectin was evidenced by the occurrence of a strong band at 1745 cm^{-1} , also occurring in the blend, ascribed to esterified groups, and at 1624 assigned to free carboxyl groups⁴⁸. The bands occurring at 1105 and 1010 cm^{-1} , in the finger-print region are typical of pectin polymers and can be attributed to the C=O stretching^{31, 49}.

In the DNA and chitosan-DNA gels it is possible to observe a strong band at 1650 cm^{-1} , slightly shifted from the chitosan band, assigned to the C2=O2 stretching vibration of cytosine⁴³; the shoulder found at the blend and at DNA gels, at 1590 cm^{-1} , can be attributed to a couple of C6=O6, C5-C6, and C4=C5 stretching vibrations of guanine⁵⁰. The small modifications of the FTIR spectra around 1590 cm^{-1} may suggest a different base-pairing pattern of DNA alone and in the blend⁴³. Further bands, in the CHD spectrum, at 526 cm^{-1} and 890 cm^{-1} , the later only observed in the DNA-containing gels, indicate the deoxyribose region⁵¹, and the deoxyribose-phosphate groups⁵².

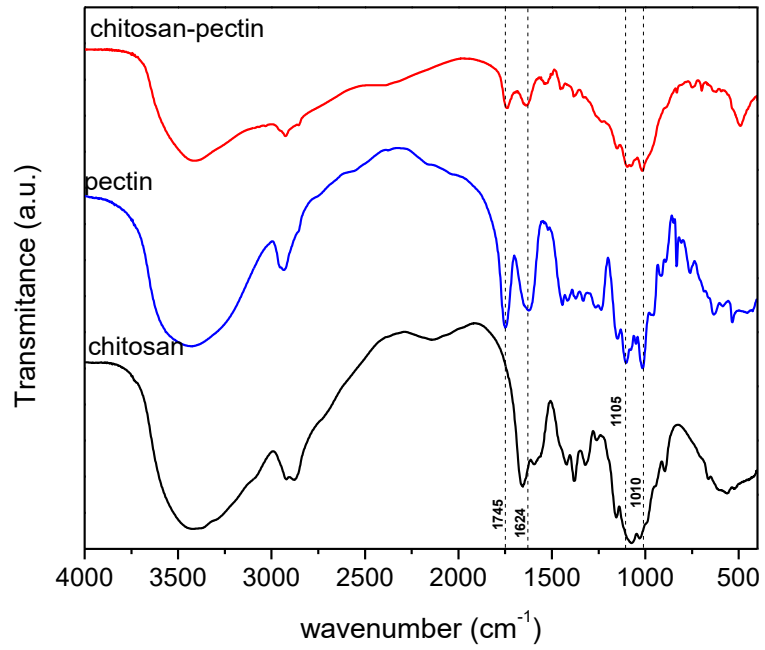


Figure 1. Infrared spectra of chitosan, pectin and the chitosan-pectin blend gel.

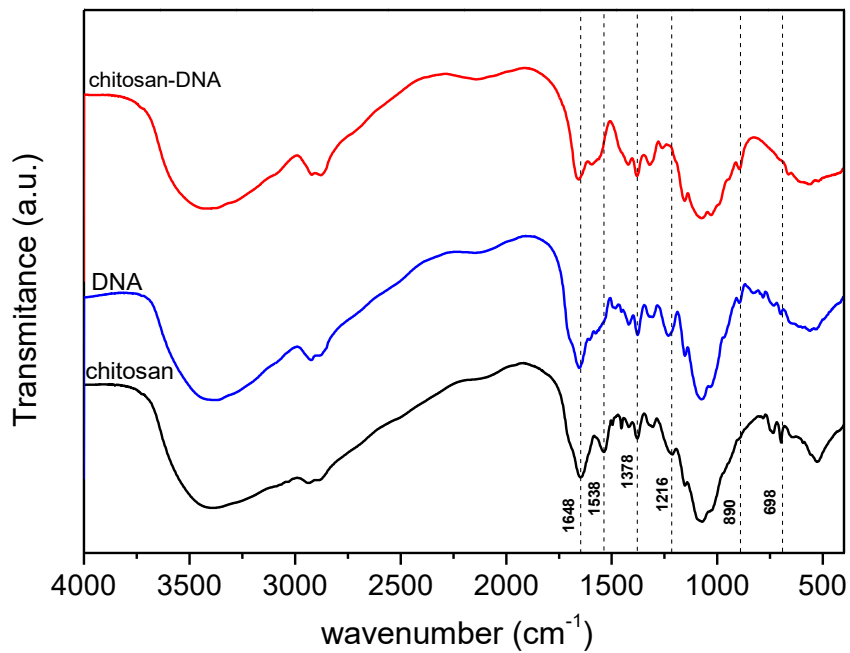


Figure 2. Infrared spectra of chitosan, DNA and the chitosan-DNA blend gel.

The gels have also been characterized by thermogravimetric analysis (Figures 3 and 4). To have an assessment on the main degradation temperatures, the maximum degradation rate has been computed (dTG), and the corresponding temperature, T_{max} , obtained. In the Figure 3, and excluding the T_{max} obtained at temperatures below 150 °C, caused by the evaporation of residual water, two main degradation steps at 234 °C and 292 °C were found for pectin^{53, 54} and chitosan⁵⁵, respectively. It is interesting that for the chitosan-pectin blend two degradation steps can be found at 227 °C and a 548 °C; the latter can be justified by the degradation of chitosan. The former maximum degradation temperature, corresponding to ca. 63% weight loss, occurs at temperature below the main transition steps for single pectin and chitosan. This can be justified by a decrease in the thermal stability of blend as a consequence of ionic interactions between pectin and chitosan, leading to a plasticizer-like effect.

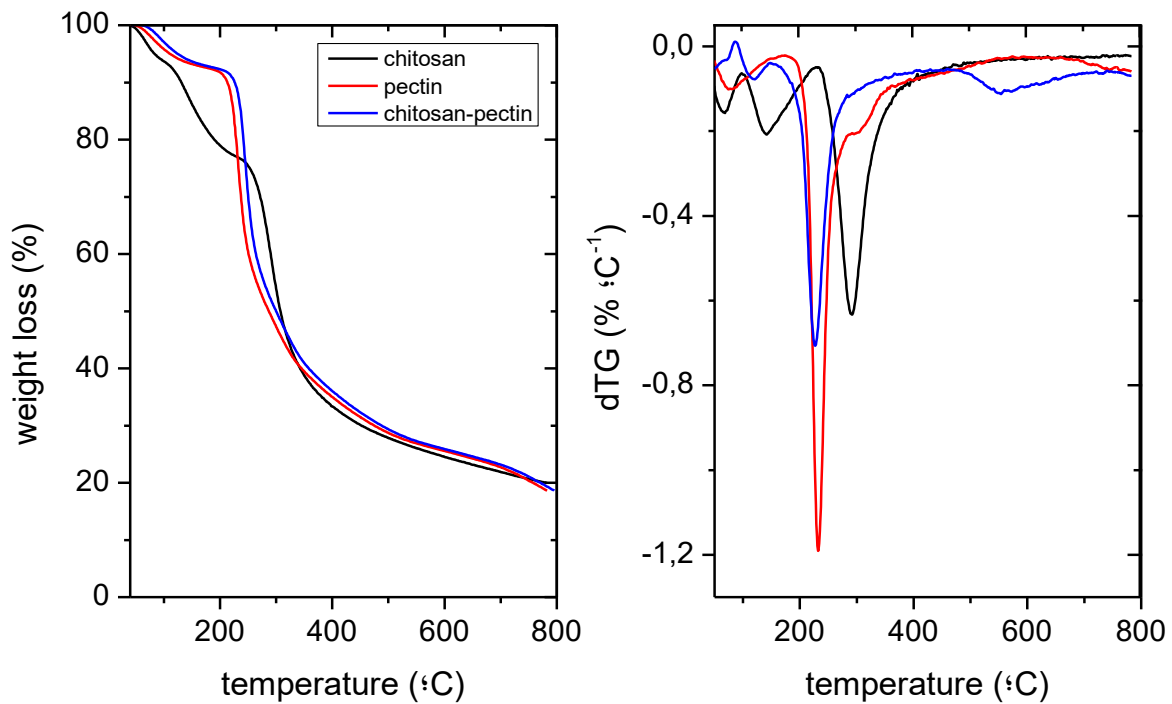
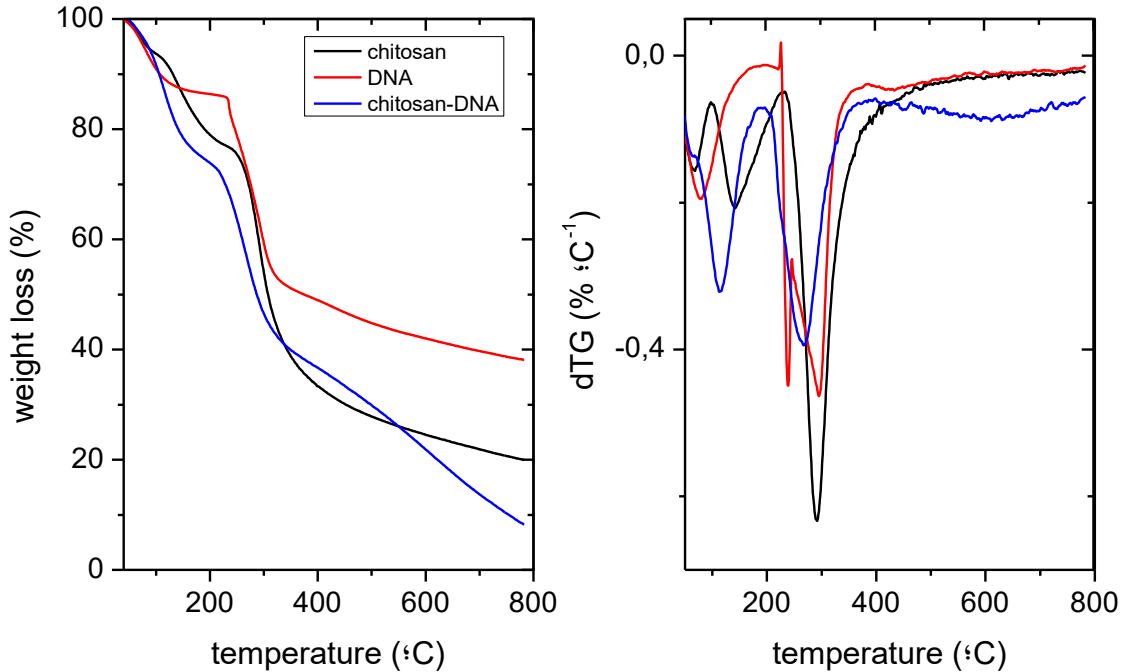


Figure 3. TGA and dTG curves of CS (—), Pec (—) and CS-Pec (—).

Figure 4. TGA and dTG curves of chitosan (—), DNA (—) and chitosan-DNA (—).



The analysis of figure 4 shows that the first degradation step of DNA occurs at 238 °C⁵⁶, followed by a second degradation step at 295 °C, suggesting that the presence of different secondary structures might be present. However, the most remarkable difference between this blend, and the chitosan-pectin one, is that the main degradation step occurs at 268 °C, suggesting a higher degree of phase separation.

From the analysis of the surface morphology of chitosan-pectin and chitosan-DNA gels (Figure 5), it can be observed that the surface of the latter shows a more heterogeneous structure, characterized by well-defined aggregates and fiber-like structures. On the other hand, for chitosan-pectin the surface morphology is more featureless, non-porous structure, indicating some irregularities that might be related with some phase separation. This morphology can increase stability by providing higher resistance⁵⁷. However, after the GP sorption a significant change in the surface morphology of

chitosan-DNA is observed; i.e., the surface becomes highly porous. This has been reported for other DNA-containing composites, such as PVA-DNA⁴³, and suggests that the GP has a significant effect or even interaction with the polymers. This might also justify the lowest amounts of GP release by chitosan-DNA (see the discussion below), when compared with the chitosan-pectin, despite its higher porosity. However, the surface of chitosan-pectin does not undergo relevant modification upon GP loading, indicating that the pectin is playing a major role⁵⁸.

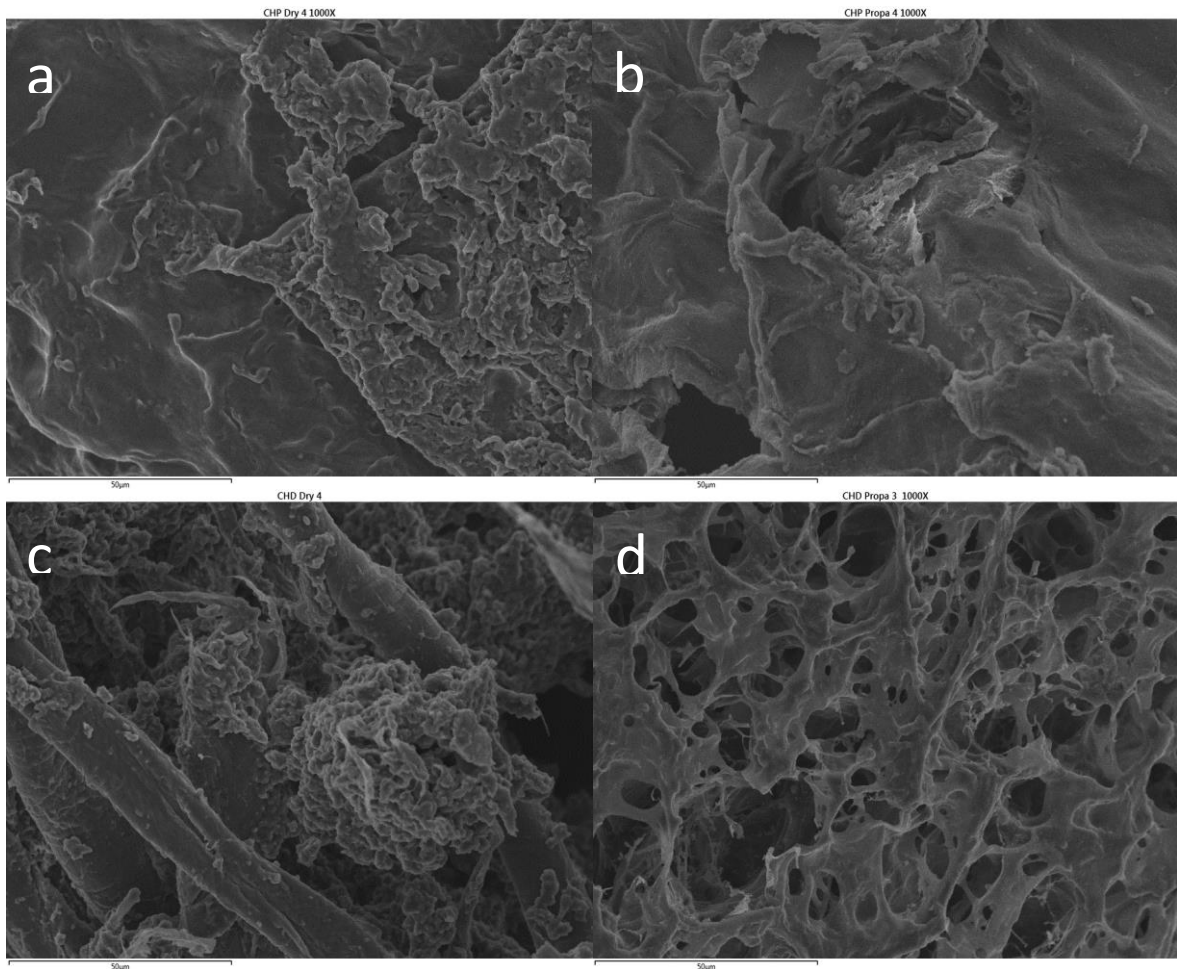


Figure 5. Scanning electron micrographs of chitosan-pectin (a, b) and chitosan-DNA (c, d) before (a, c) and after (b, d) GP loading. $\times 1000$ magnification. Scale = 50 μ m.

Loading and Drug release

Table 1 shows the encapsulation efficiency of GP into chitosan-pectin and chitosan-DNA blend gels. It can be seen that the EE in the chitosan-DNA is significant lower (ca. 3.5 fold lower) than those obtained for chitosan-pectin. This might be related with the occurrence of higher phase separation in the chitosan-DNA gel and also the lower swelling degree of this gel ($Q=700\%$) when compared with that for chitosan-pectin ($Q=1109\%$). However, it is surprising that the cumulative release, in percentage, is higher for the chitosan-DNA gel. This might be due to the change in the structure of pectin-DNA gel, becoming a more porous surface structure (see Figure 5). In any case, the absolute value of the maximum released amount of GP is always higher for chitosan-pectin.

Table 1. Encapsulation efficiency and cumulative release of GP in chitosan-pectin and chitosan-DNA coacervate gels, at 25°C.

	Chitosan-pectin		Chitosan-DNA	
	25 °C	37 °C	25 °C	37 °C
	Water			
EE (%)	19	21	7	8
Cumulative release (%)	13.2	12.1	25.7	30
	simulated saliva			
EE (%)	30	38	8	10
Cumulative release (%)	13.9	19.9	31.6	36.5

Figures 6 and 7 show the release profiles of Green Propolis from chitosan-pectin and chitosan-DNA gels, respectively, to simulated saliva at 25 and 37 °C.

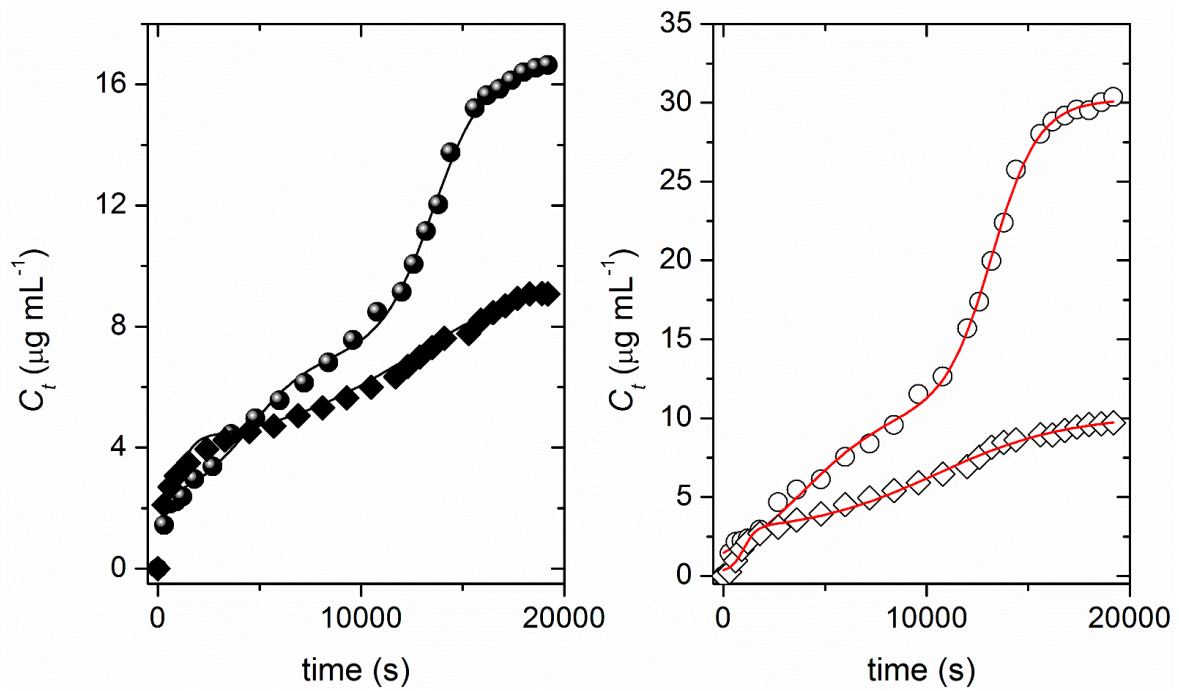


Figure 6. Release kinetics of GP from chitosan-pectin gels to water (diamond) and simulated saliva (spheres), at 25 and 37 °C (black and white data points, respectively). Solid lines were obtained by fitting eq. (3) to the experimental data.

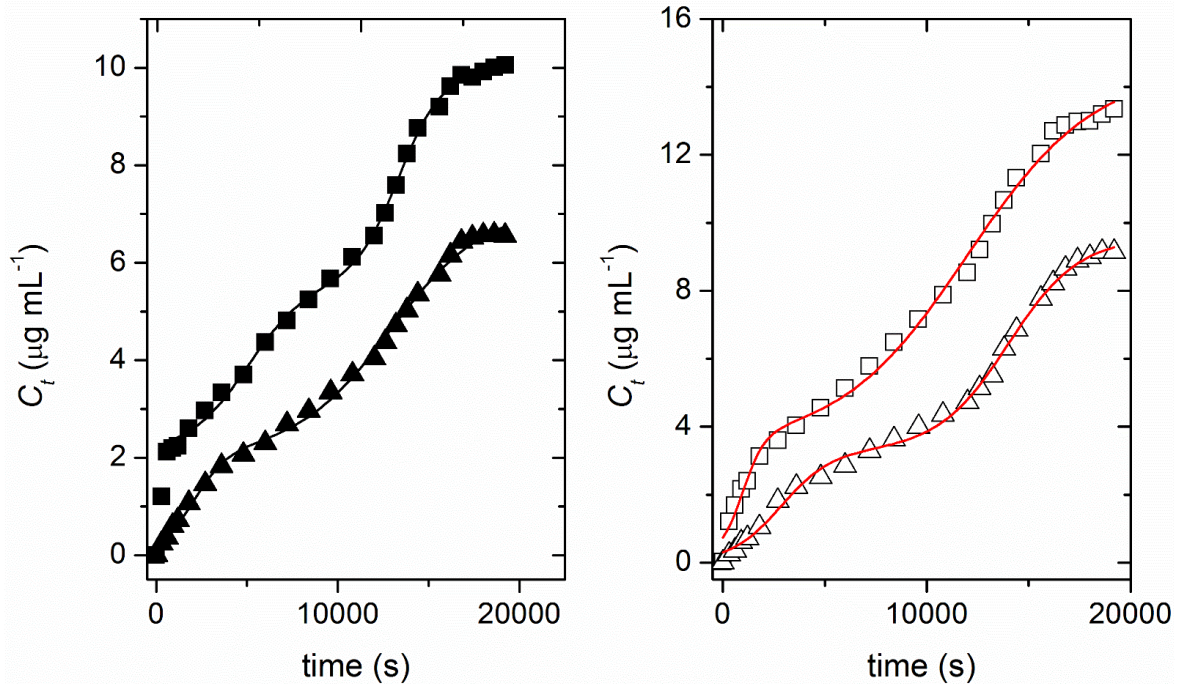


Figure 7. Release kinetics of GP from chitosan-DNA gels to water (triangle) and simulated saliva (square), at 25 and 37 °C (black and white data points, respectively). Solid lines were obtained by fitting eq. (3) to the experimental data.

As a first approach we can observe that in all cases the release kinetics does follow a two-step mechanism; however, it is also clear that such a mechanism occurs concomitantly once no plateau is observed just before the beginning of the 2nd significant release – this will be discussed later. Two other points are worth noticing: the cumulative release increase by increasing the temperature (the exception occurs for the release from chitosan-pectin to water), being more significant in the case of chitosan-pectin gel; and at 25 °C a boost release can be observed during the first 10 minutes release. The later process might be justified by a surface phenomenon (either surface erosion or preferential adsorption)⁵⁹; however, that process is somewhat hindered by increasing the temperature and, consequently, the origin of mass transport becomes not relevant.

In order to have a better insight on the release mechanism several heuristic models have been tested (e.g., Weibull, Logistic, Korsmeyer-Peppas and Peppas-Sahlin⁶⁰⁻⁶² without success – results not shown. However, assuming that the release is characterized by a biphasic release that may occur simultaneously, as can be inferred from the continuous increase of the GP as a function of time (i.e., no two plateaus of cumulative release concentration as a function of time are observed), the following equation has been used⁶³:

$$C_t = C_0 + (C_{eq} - C_0) \left(\frac{p}{1 + 10^{(\log t_1 - t)k_1}} + \frac{1-p}{1 + 10^{(\log t_2 - t)k_2}} \right) \quad (3)$$

where C_0 , C_t and C_{eq} are the initial, cumulative and equilibrium release concentrations of GP, respectively, $\log t_1$ and $\log t_2$ correspond to the first and second EC_{50} (i.e. the concentrations of the drug that gives half-life response) and p is the contribution of each phase for the whole release profile. All the fitting parameters, summarized in the Table 2, have been estimated through a least-squares method, with a confidence degree of 95%, by using Origin 8.5 software, and keeping C_0 as constant and equal to zero.

Table 2. Fitting Parameters of eq.3 to experimental release data of GP Loaded into chitosan-pectin and chitosan-DNA gels to simulated saliva, at 25 and 37 °C.

	Chitosan-pectin		Chitosan-DNA	
	25°C	37°C	25°C	37 °C
	<i>Water</i>			
$C_{eq}/(\square g mL^{-1})$	10.1 (± 0.4)	10.2 (± 0.3)	7.2 (± 0.2)	9.6 (± 0.2)
$logt_1 /s$	1051 (± 73)	1005 (± 74)	2051 (± 203)	2851 (± 214)
$logt_2 /s$	12571 (± 390)	10460 (± 360)	12878 (± 242)	14005 (± 158)
k_1 /s^{-1}	$1.2 (\pm 0.3) \times 10^{-3}$	$1.5 (\pm 0.5) \times 10^{-3}$	$4.7 (\pm 0.2) \times 10^{-4}$	$3.5 (\pm 0.4) \times 10^{-4}$
k_2 /s^{-1}	$1.2 (\pm 0.2) \times 10^{-4}$	$1.3 (\pm 0.2) \times 10^{-4}$	$1.7 (\pm 0.2) \times 10^{-4}$	$2.5 (\pm 0.2) \times 10^{-4}$
P	0.24 (± 0.04)	0.26 (± 0.04)	0.29 (± 0.04)	0.34 (± 0.02)
R^2	0.9963	0.9952	0.9970	0.9972
	<i>Simulated saliva</i>			
$C_{eq}/(\square g mL^{-1})$	16.7 (± 0.2)	30.2 (± 0.3)	10.1 (± 0.6)	14.6 (± 0.6)
$logt_1 /s$	4468 (± 363)	3948 (± 529)	4906 (± 300)	969 (± 127)
$logt_2 /s$	13567 (± 131)	13225 (± 118)	13379 (± 147)	11972 (± 321)
k_1 /s^{-1}	$2.9 (\pm 0.4) \times 10^{-4}$	$2.0 (\pm 0.3) \times 10^{-4}$	$2.7 (\pm 0.3) \times 10^{-4}$	$8 (\pm 2) \times 10^{-4}$
k_2 /s^{-1}	$3.3 (\pm 0.3) \times 10^{-4}$	$3.8 (\pm 0.4) \times 10^{-4}$	$3.3 (\pm 0.4) \times 10^{-4}$	$1.4 (\pm 0.2) \times 10^{-4}$
P	0.34 (± 0.03)	0.36 (± 0.03)	0.43 (± 0.04)	0.24 (± 0.04)
R^2	0.9974	0.9978	0.9981	0.9951

The values inside parenthesis are the standard deviations.

From the analysis of Figures 6 and 7 and C_{eq} summarized in the table 2, it can also be concluded that the amount of GP released is significantly higher in synthetic saliva than in water. The amount of GP released reaches a value between 14 and 36% of the encapsulated amount of GP. This may be justified by the ionic strength effect on these gels. Once both gels are ampholytic gels, an increase in the ionic strength may lead to a swelling effect which may stabilize both polymers^{64, 65} and thus contributing for an enhancement of GP release. This is also followed by a significant increase in the EC50 for the first step

release (*logt1*), comparing with those observed for water system. A final remark concerns the contribution of both steps for the whole release process. In all systems, the first step mechanism contributes with ca. 30 % for the total release of GP, suggesting that the main release process is probably affected by some polymeric relaxation processes, further facilitating the mobility and release of the GP⁶⁶⁻⁶⁸.

Biological Analysis

For the biological analysis, the results are showed in Figure 8.

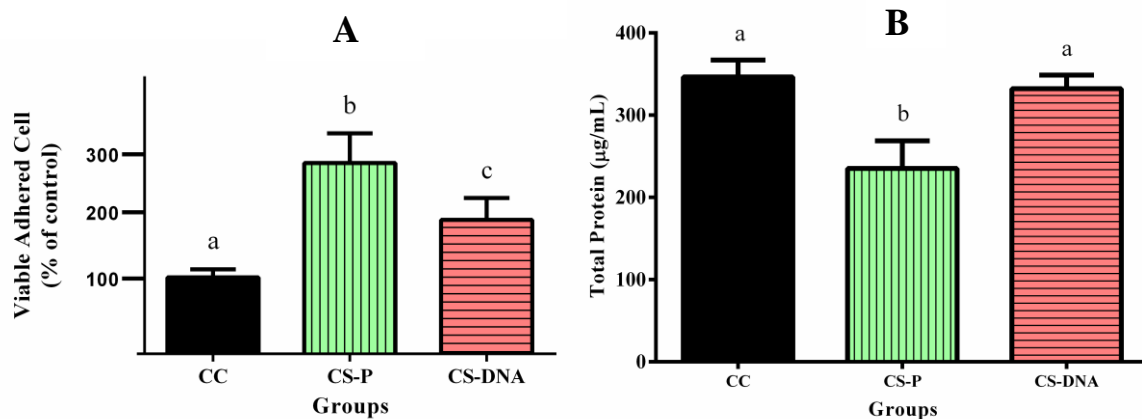


Figure 8- Graphics reporting mean \pm SD for in vitro tests: a) measurement of cell viability MTT assay; b) Total Protein production. Values that do not share the same superscript letters are significantly different from each other ($p < 0.05$).

The cell viability was obtained by MTT assay, and measurement of cell viability of experimental groups were compared to control group (100%) (Figure 8A). The CS-P and CS-DNA influenced positively in the cell culture and improved the number of viable cell. However CS-P experimental group exhibited statistical difference with control group ($p < 0.0001$), which presented

the lower cell viability. This result is in accordance with previous studies, which was observed increase of viable cells in different lineage cells when the Chi was used as biomaterial base⁶⁹⁻⁷¹. The rates of cell viability exhibited by both groups make them suitable for biomedical use according to the International Organization for Standardization (ISO) and national standards for Biological Evaluation of Medical Devices, ISO 10993-5⁷².

Regarding to total protein (Figure 8B), there was no statistical difference between control group and CS-DNA ($p>0.001$), but the values from CS-P experimental group were lower than control group, with statistical difference ($p=0.0002$).

The results exhibited represents that these new biomaterials are acceptable in the biological environment and can provide evidence for the potential of biomimetic material for tissue engineering.

CONCLUSION

It can be concluded that the hydrogel complexation of Chitosan with DNA and Pectin is a viable alternative to drug delivery system of herbal medicine such as GP.

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3 CONSIDERAÇÕES GERAIS

Sob os aspectos atuais do desenvolvimento de biomateriais, a proposição do presente estudo delineou a produção de duas diferentes matrizes à base de um biopolímero já utilizado para diferentes aplicações biomédicas, a quitosana. (Berger et al., 2005; Cheung et al., 2015).

A primeira matriz desenvolvida, um arcabouço de fibras ultrafinas eletrofiadas com e sem inclusão de cristais de nanohidroxiapatita se comportou como um material biomimético, promovendo a diferenciação celular. E especificamente em relação à diferenciação osteoblástica, a associação de hidroxiapatita e polímeros apresenta resultados positivos para o metabolismo celular (Wutticharoenmongkol et al., 2007). Assim, neste estudo, ChHa proporcionou maiores médias de atividade de fosfatase alcalina, sendo que este é um indicador precoce da diferenciação dos osteoblastos e sua expressão é considerada um parâmetro funcional *in vitro*, refletindo o progresso da diferenciação celular (Hoemann et al., 2009).

A segunda matriz desenvolvida, um hidrogel, foi subdividida em um hidrogel de quitosana coacervado com DNA de salmão e um hidrogel de quitosana coacervado com Pectina. Ambas apresentaram potencial para liberação controlada de fármacos. Pode-se observar que, em todos os casos estudados, a cinética de liberação segue um mecanismo de duas etapas que ocorrem concomitantemente, já que nenhum platô é observado. As duas etapas contribuem para todo o processo de liberação. Em todos os sistemas, o mecanismo da primeira etapa contribui com 30% para a liberação total do GP, sugerindo que o processo principal de liberação provavelmente é afetado por alguns processos de relaxamento polimérico, facilitando ainda mais a mobilidade e a liberação do GP (Faisant et al., 2002; Polishchuk, Zaikov, 1997;

Siepmann et al., 1999).

Assim sendo, considerando as limitações metodológicas empregadas, foi possível desenvolver e caracterizar dois materiais com potenciais de atuação complementar.

Sugere-se com isso, que as possíveis aplicações de ambos materiais podem se delinear da seguinte maneira: clinicamente, a matriz de fibras poderá atuar como uma membrana de regeneração óssea guiada, para posterior instalação de implante osseointegrado, e o hidrogel poderá ser utilizado como um mecanismo de terapia profilática, utilizado entre os componentes das próteses sobre implantes, evitando a proliferação microbiana ou mesmo na superfície de implantes no tratamento da periimplantite.

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ANEXO A – Comprovante de artigo enviado para publicação

16/04/2018

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Manuscript ID
ADJ-04-18-0198

Title
Potential of nanohydroxyapatite loading on electrospun chitosan fibers: biological and morphological studies


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16-Apr-2018

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ANEXO B – Certificado do Comitê de Ética em Pesquisa


UNIVERSIDADE ESTADUAL PAULISTA
"JÚLIO DE MESQUITA FILHO"
 Campus de São José dos Campos
 Instituto de Ciência e Tecnologia


CERTIFICADO
CEUA – Comissão de Ética no
Uso de Animais

CERTIFICAMOS, que o protocolo registrado sob o nº 01/2017, intitulado:- **"Eletrofiliação via solução de fibras ultrafinas de quitosana com nanohidroxiapatita: caracterização física e biológica"** sob a responsabilidade de **LUANA MAROTTA REIS DE VASCONCELLOS**, tendo como colaboradora :- Tábata do Prado Sato, e que envolve a utilização de animais pertencentes ao filo Chordata subfilo Vertebrata (exceto humanos), para fins de pesquisa científica, encontra-se de acordo com os preceitos da Lei nº 11.794, de 8 de outubro de 2008, do Decreto nº 6.899 de 15 de julho de 2009 e com as Normas editadas pelo Conselho Nacional de Controle de Experimentação Animal (CONCEA), e foi **APROVADO** pela **COMISSÃO DE ÉTICA NO USO DE ANIMAIS (CEUA – ICT – CAMPUS DE SÃO JOSÉ DOS CAMPOS-UNESP)**, em reunião de 07/04/2017.

Finalidade	() Ensino (X) Pesquisa Científica
Vigência da Autorização	07/04/2017 a 20/12/2018
Espécie/linhagem/raça	Rato/heterogênico Wistar
Nº de Animais	20
Peso/idade	400 grs / 90 dias
Sexo	macho
Origem	Biotério Central – Campus de Botucatu-UNESP

São José dos Campos, 07 de abril de 2017


 Profa.Dra. **PAULA CAROLINA KOMORI DE CARVALHO**
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